



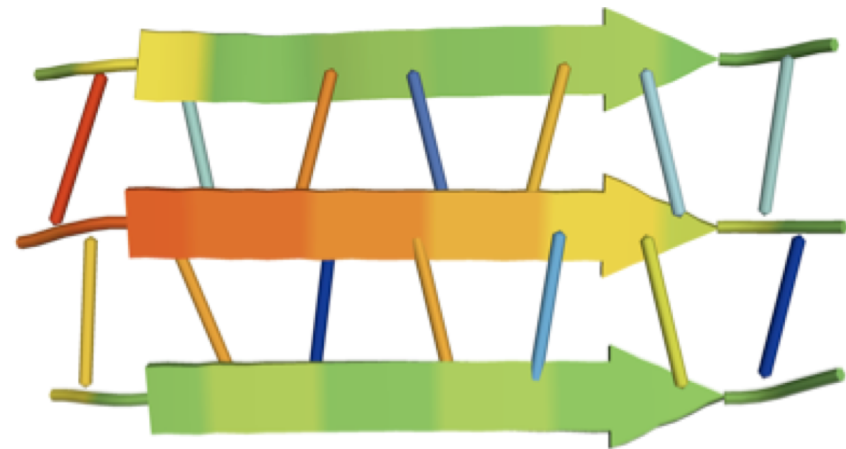
Heidelberg Institute for
Theoretical Studies



Simulations of biomaterials under force

[part I: Collagen]
part II: Silk

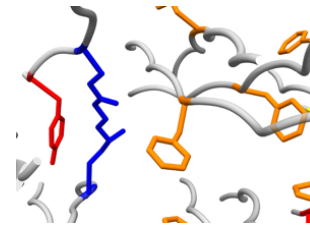
Frauke Gräter, Bogota, 09/2019



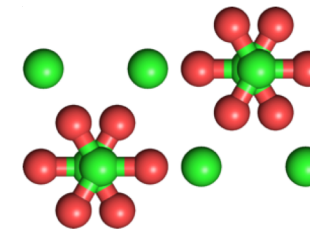
MOLECULAR (bio)mechanics – why?



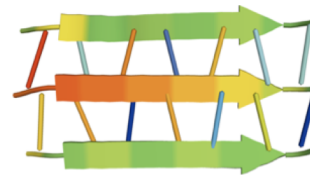
collagen



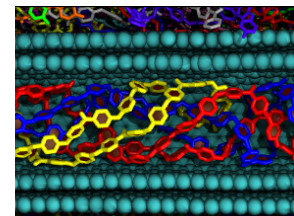
mineralized tissue



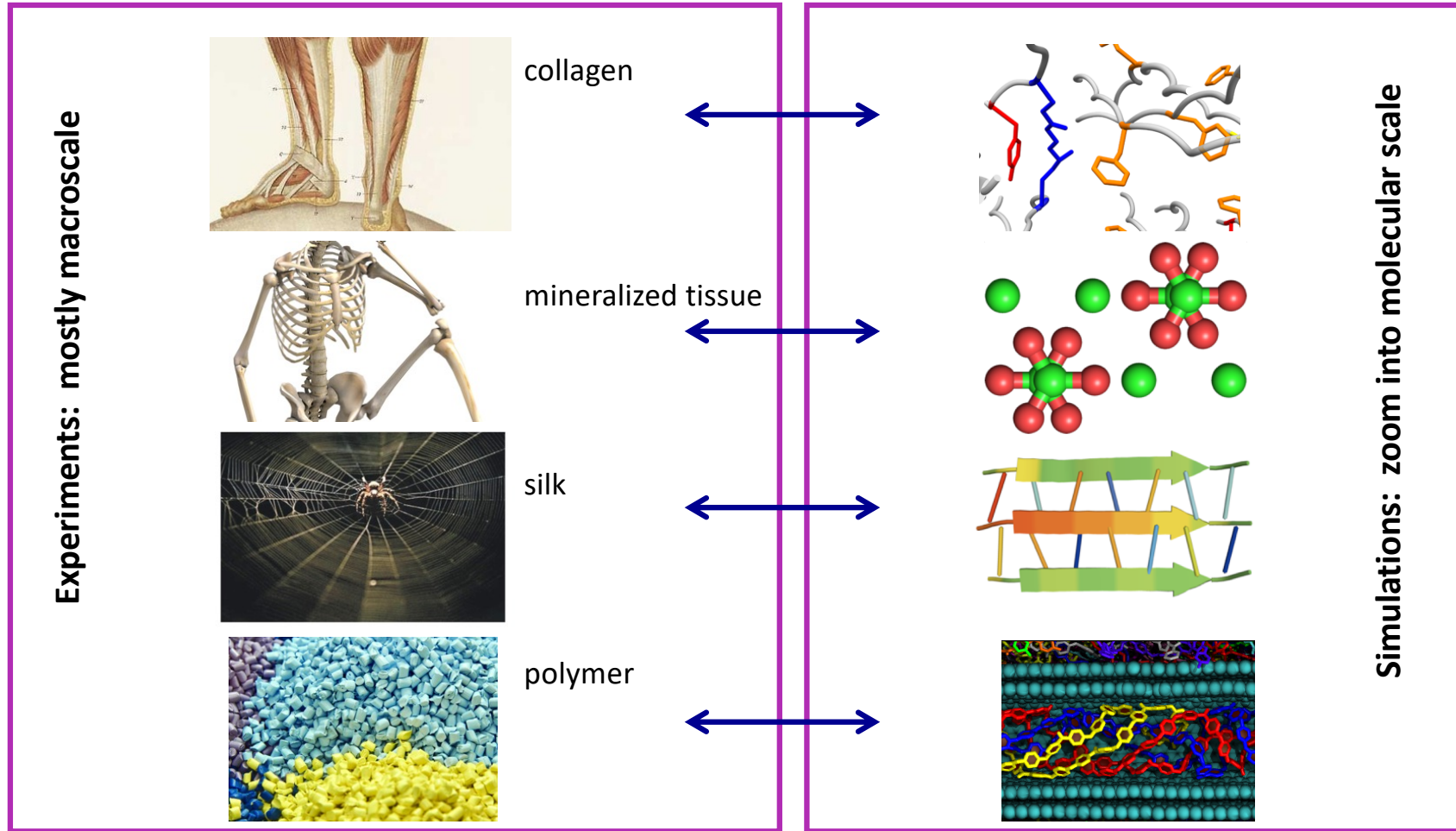
silk



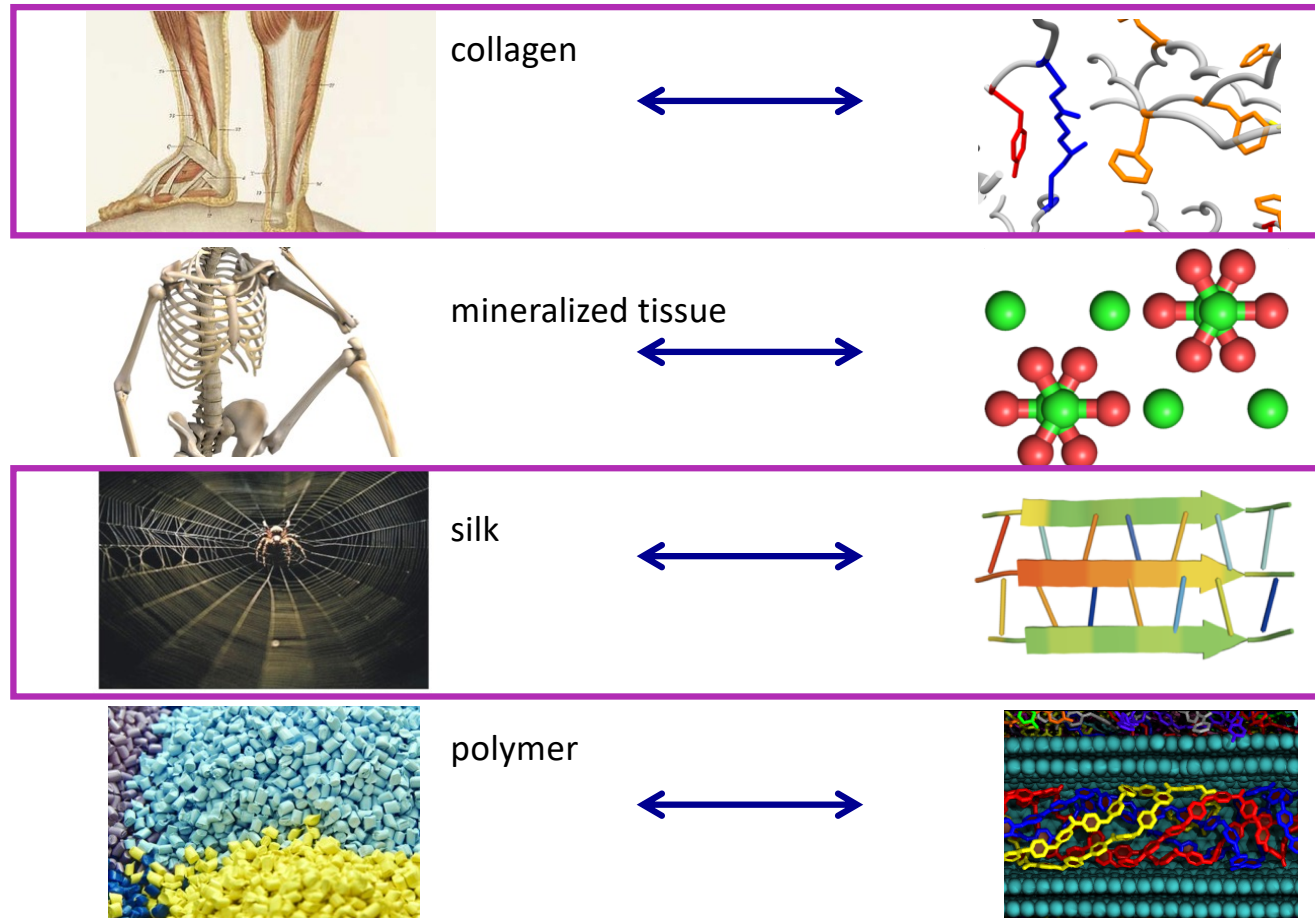
polymer

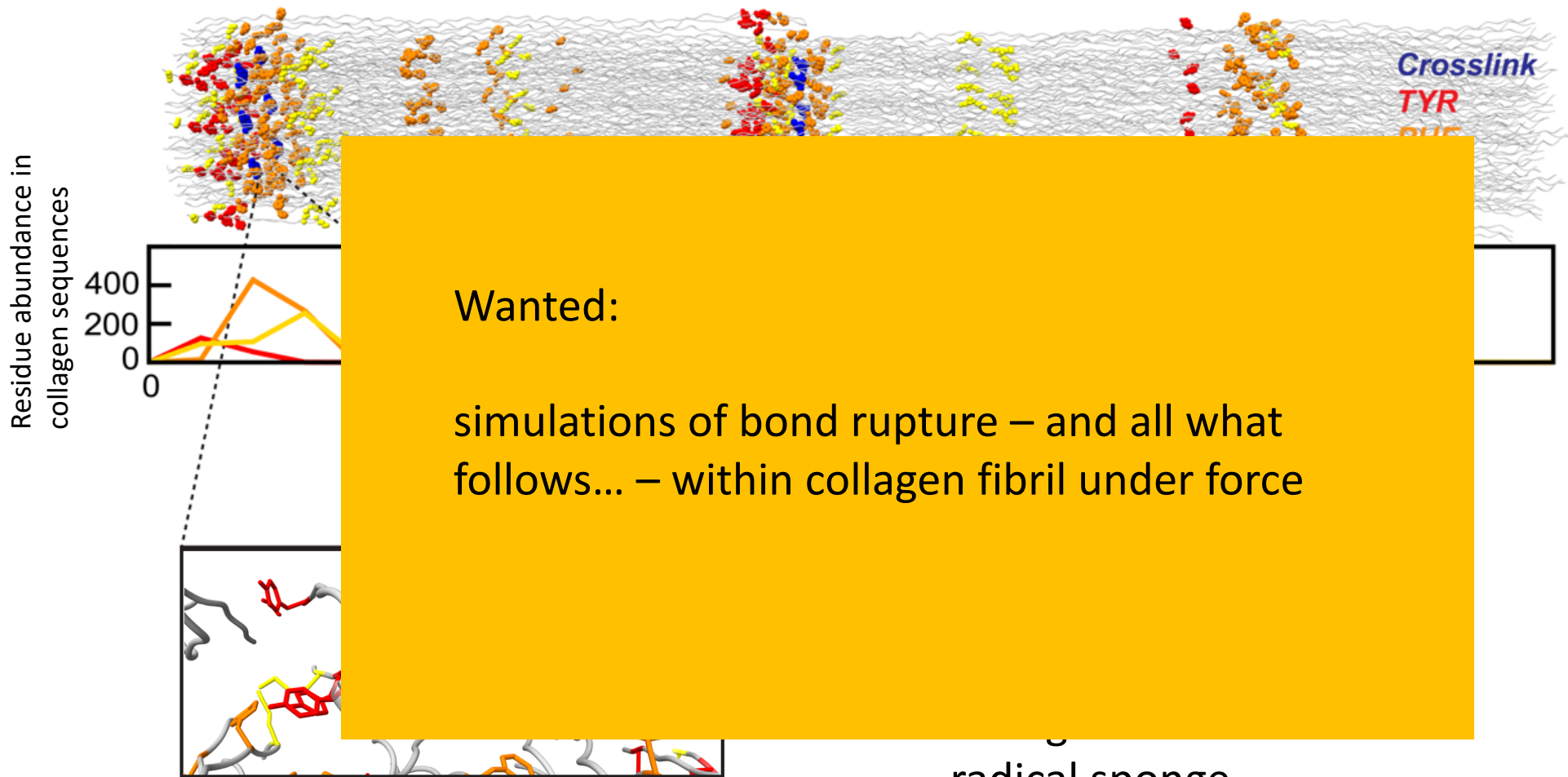


MOLECULAR (bio)mechanics – why?



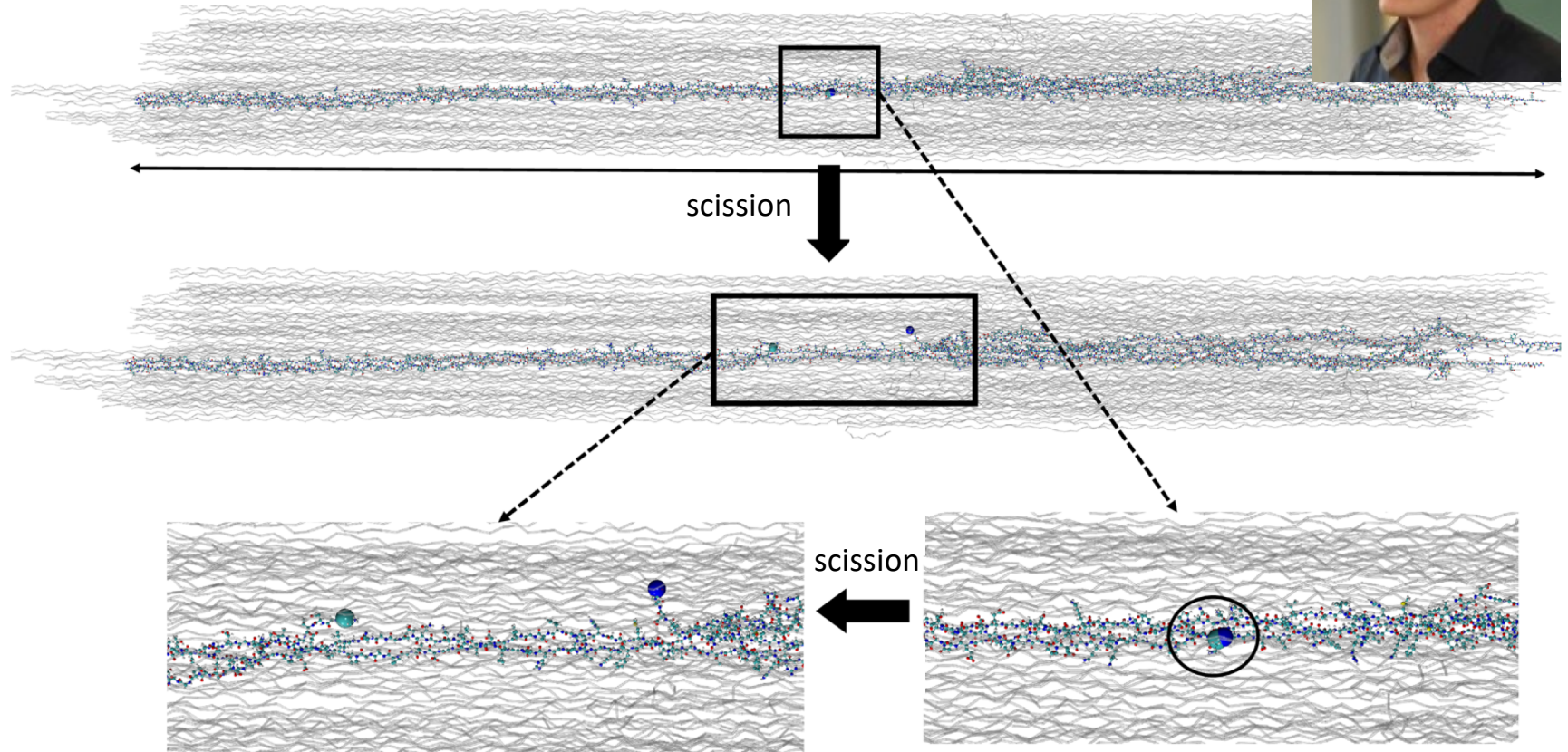
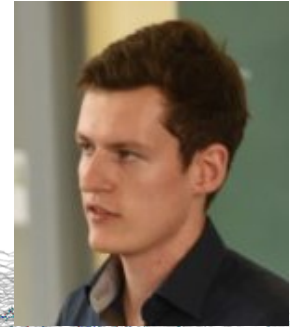
MOLECULAR (bio)mechanics – why?



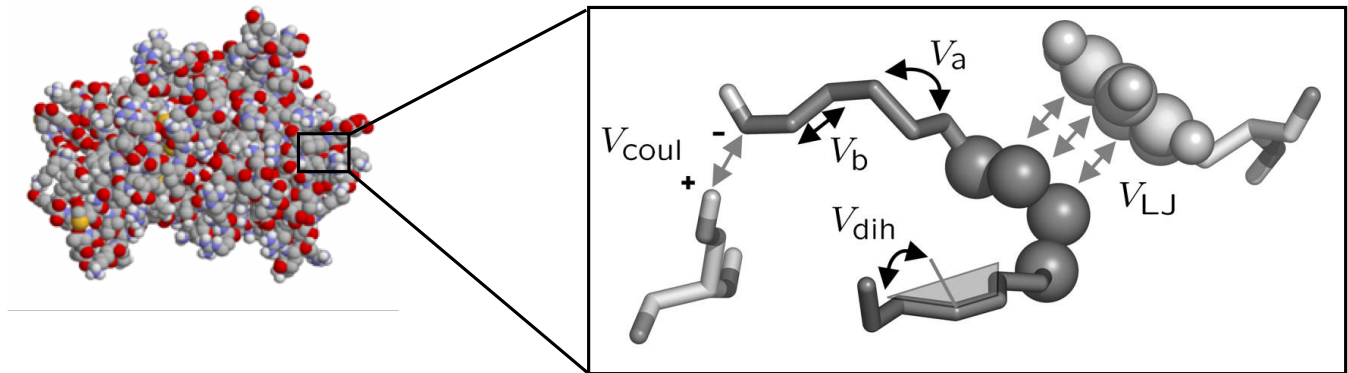


Reactive Molecular Dynamics: spontaneous chemical events on-the-fly

Benedikt
Rennekamp



Molecular Dynamics simulations: bonds can not rupture!



1.

$$\begin{aligned}
 E = & \sum_{\text{bonds}} \frac{k_i}{2} (l_i - l_{i,0})^2 \\
 & + \sum_{\text{angles}} \frac{k_i}{2} (\theta_i - \theta_{i,0})^2 \\
 & + \sum_{\text{torsions}} \frac{V_n}{2} (1 + \cos(n\omega - \gamma)) \\
 & + \sum_{i=1}^N \sum_{j=i+1}^N \left(4\epsilon_{ij} \left(\left(\frac{\sigma_{ij}}{r_{ij}} \right)^{12} - \left(\frac{\sigma_{ij}}{r_{ij}} \right)^6 \right) + \left(\frac{q_i q_j}{4\pi\epsilon_0 r_{ij}} \right) \right)
 \end{aligned}$$

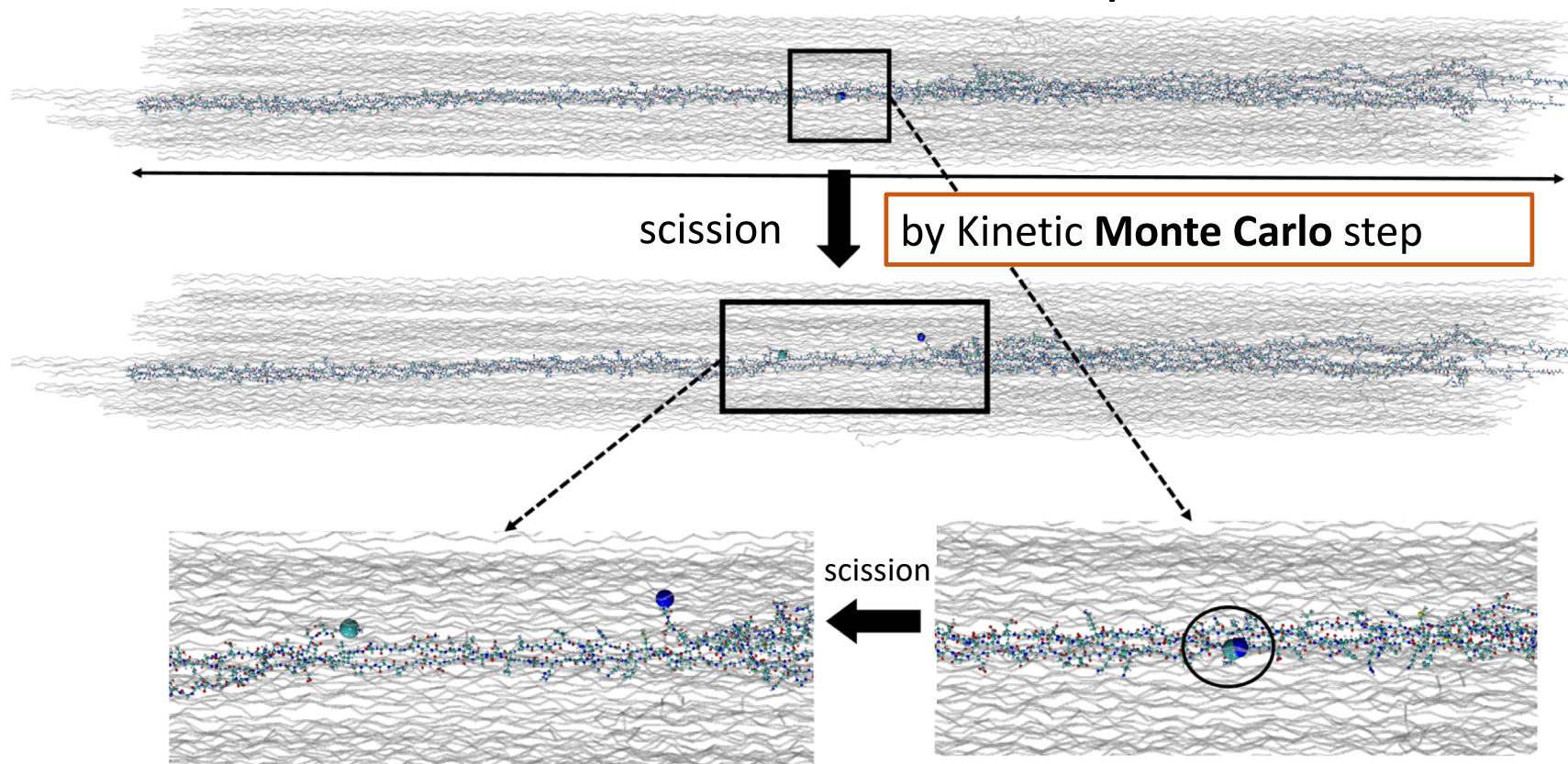
} bonded interactions \longleftrightarrow
 \longleftrightarrow
 } non-bonded interactions

2.

&

$$F_i = \frac{\delta E}{\delta r_i} = m_i \frac{d^2 r_i}{dt^2}$$

“Reactive” Molecular Dynamics: spontaneous chemical events on-the-fly



Monte Carlo methods

stochastic simulation methods, **sample randomly** to get obtain an approximate numerical result for a complex (high-dimensional?) problem

flavors:

- Monte Carlo integration
- Metropolis Monte Carlo
- Quantum Monte Carlo
- kinetic Monte Carlo

applications:

e.g. optimization, simulations of solids/molecules/polymers, weather, astro, finance, law,....



Monte Carlo methods: some history



- name was suggested by Metropolis, at Los Alamos National Laboratories for a method put forward by Fermi/Ulam
- used for calculations of neutron diffusion

Monte Carlo methods: some history

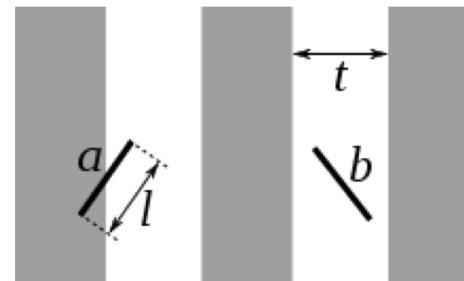
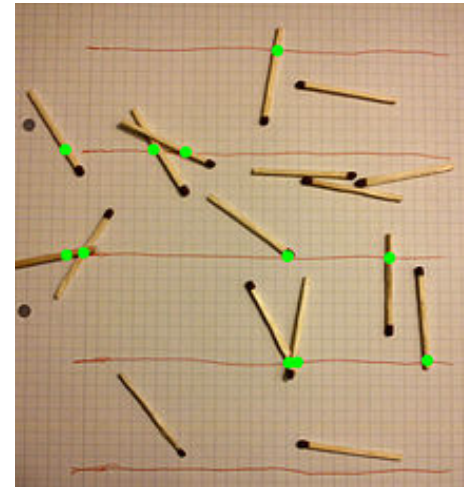
Earliest example:

Calculation of π by Laplace
(based on the Buffon Experiment)

Suppose we have a floor made of parallel strips of wood, each the same width, and we drop a needle onto the floor. What is the probability that the needle will lie across a line between two strips?

*n needles,
h of them crossing the strips
l length of needles
t width of strips*

$$\pi \approx \frac{2l \cdot n}{th}$$



wikipedia

Kinetic Monte Carlo



- Calculate different rates Γ_i
- Draw random number R between 0 and Γ_{tot}
- Carry out corresponding event
- Draw another random number T between 0 and Γ_{tot}
- Update time with $\Delta t = \log(1/T)$

Advantages:

- any time scale reachable (lower rates just lead to higher time steps)

Disadvantages:

- jump in time: no continuous trajectory
- requires good guesses for rates

“Reactive” Molecular Dynamics: spontaneous chemical events on-the-fly

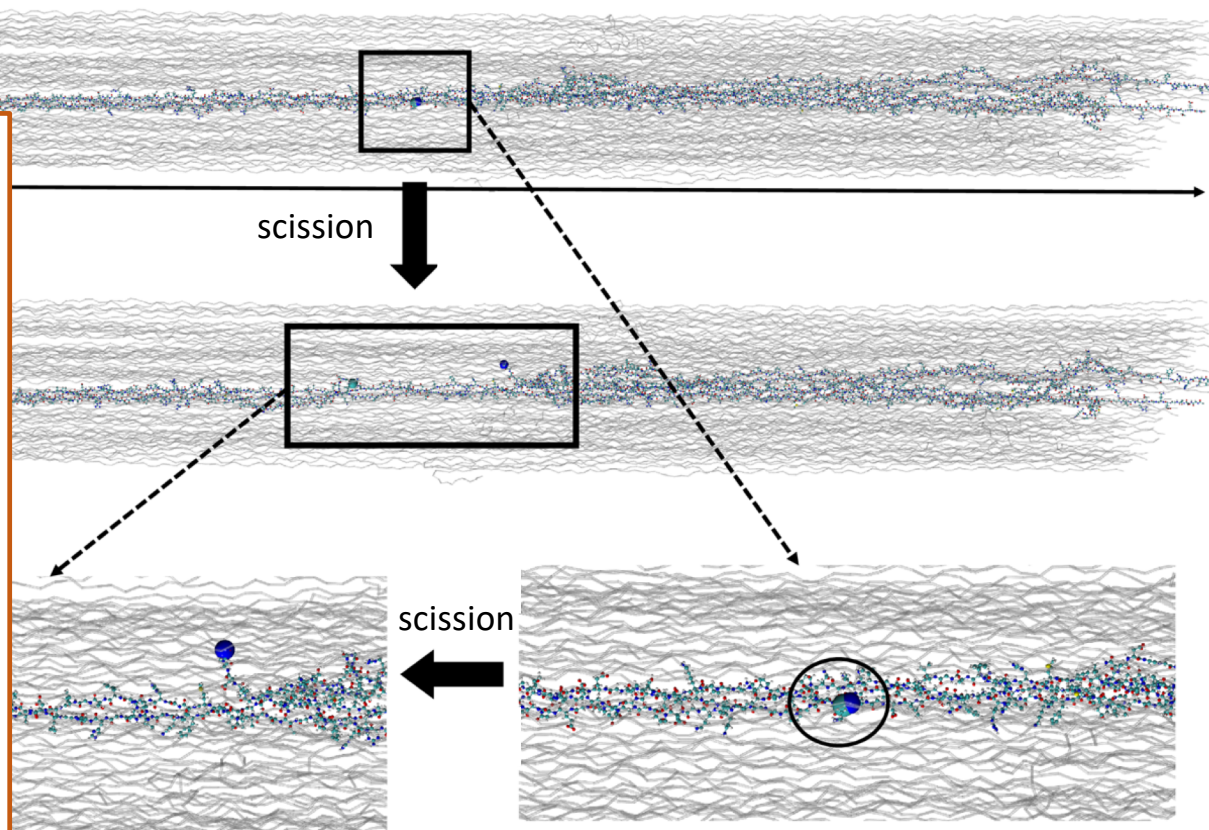
Computational scheme:

Kinetic Monte Carlo/Molecular Dynamics KIMMDY

1. Run n MD steps
2. Check forces, calculate rates
- 3. Do kinetic Monte Carlo step**
4. Perform bond scission
5. Go back to 1.

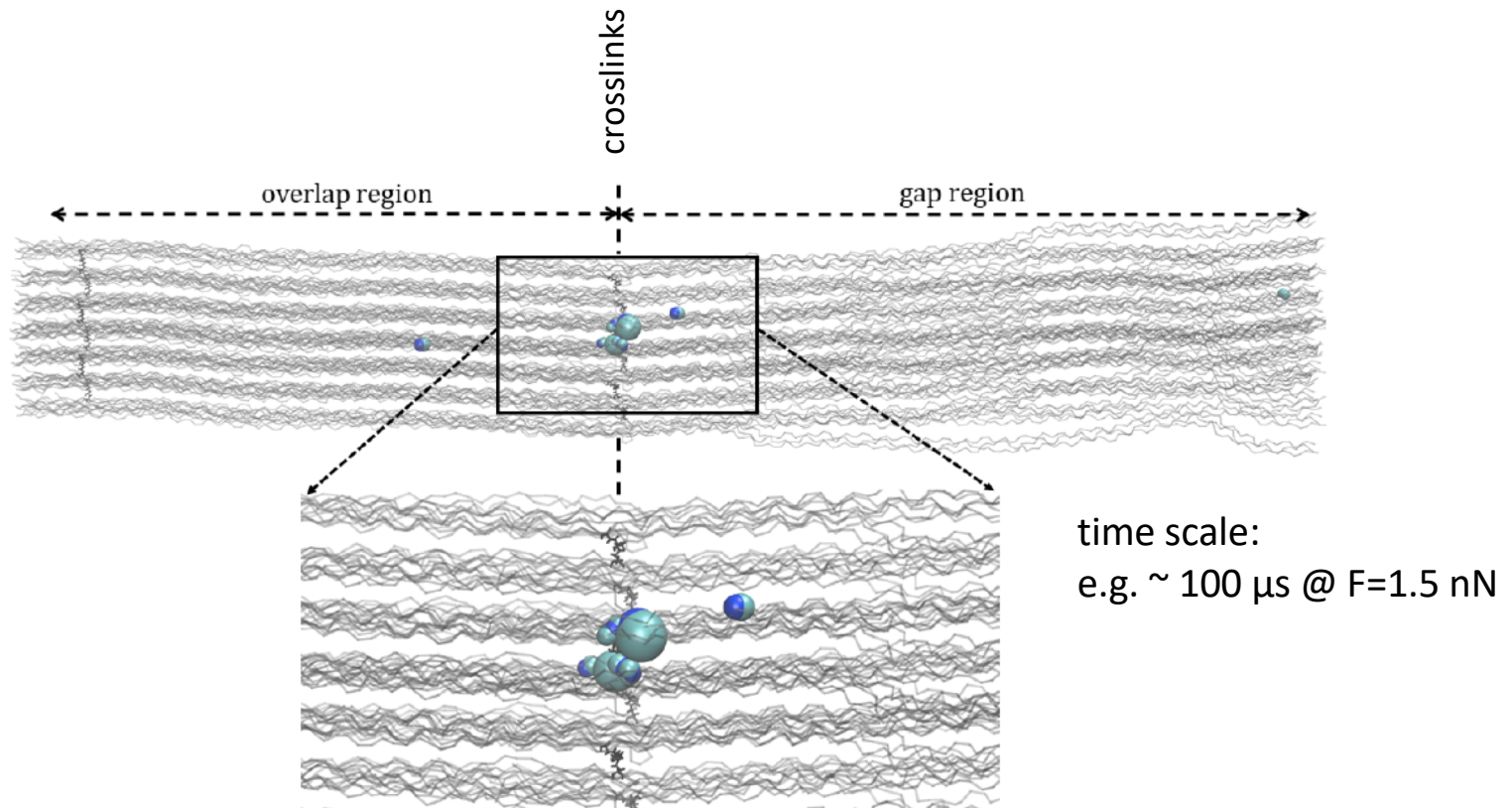
to be extended to electron transfer and
other redox-chemistries

Using **PLUMED/GROMACS**



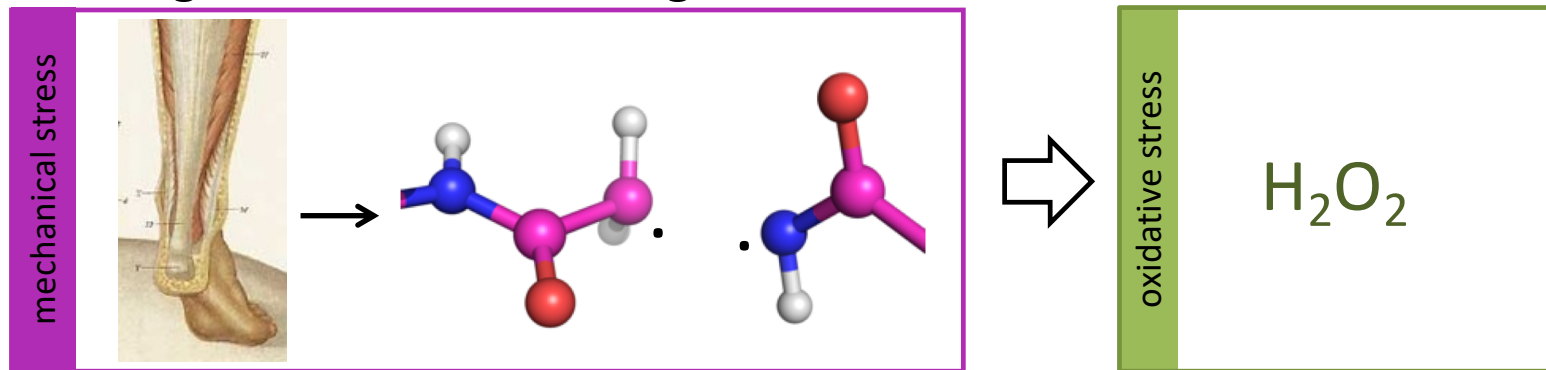
Rennekamp, Gräter, in review

It works: Rupture preferentially occurs around crosslinks



Conclusions:

YES: High loads on our collagen results in radicals



Move yourself! But only until it starts hurting...

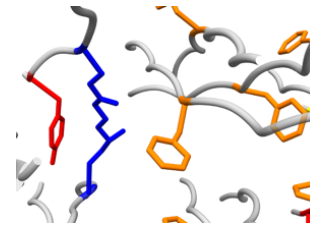


Zapp*, Obarska-Kosinska* et al, in review

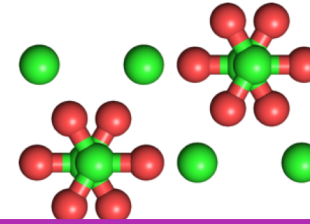
MOLECULAR (bio)mechanics – why?



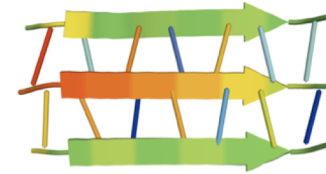
collagen



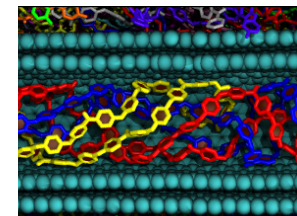
mineralized tissue



silk



polymer



Molecular
simulations
&
continuum
mechanics

Order and disorder in silk

silk molecular structure:

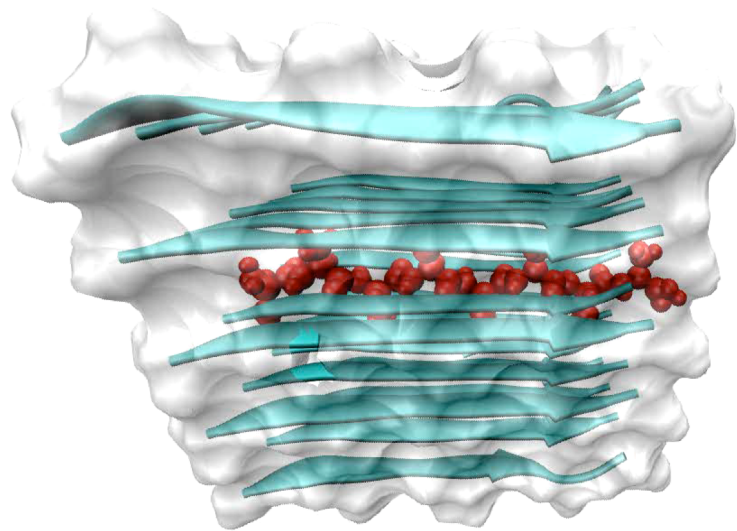


poly(alanine)



disordered
sequence
glycine rich

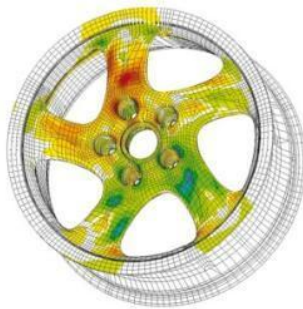




Strain and fracture: force distribution

conventional design tools:
force distribution

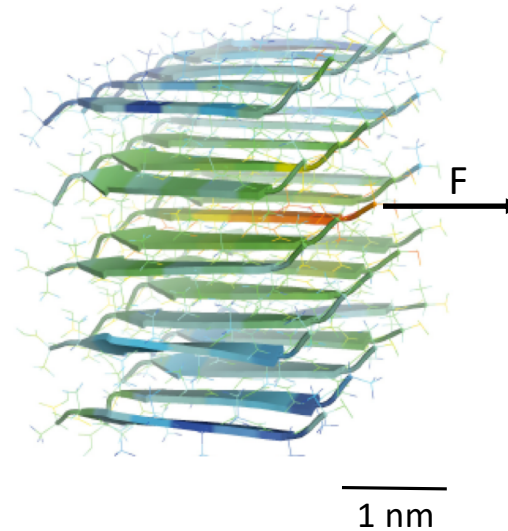
in constructions, cars ...



*macroscopic structures:
meters*

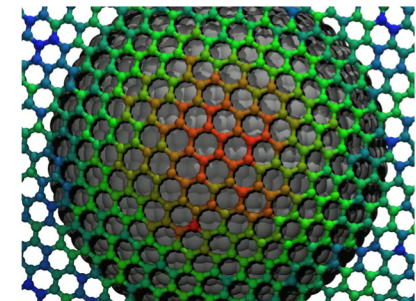
our method:
force distribution in (bio)molecules at atomistic level

e.g.



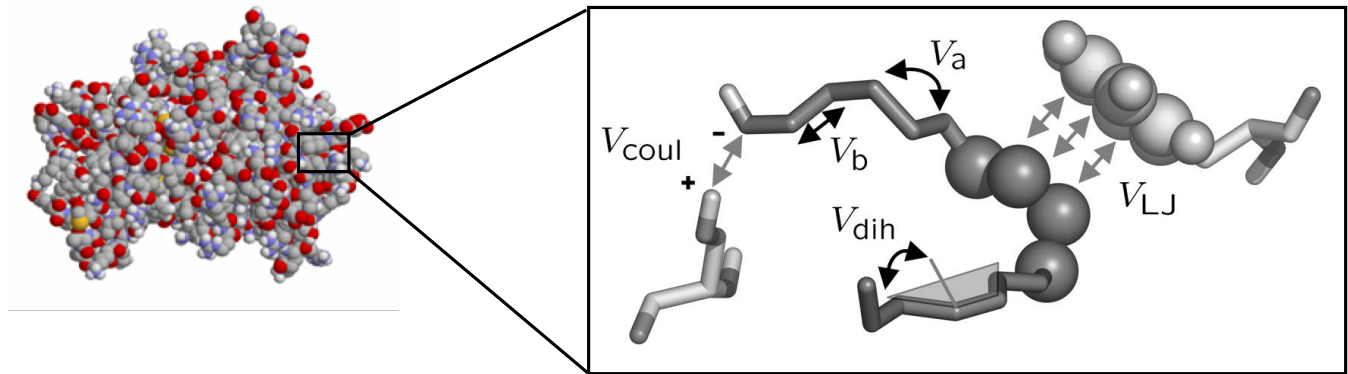
*microscopic structures:
 $\sim 10^{-9}$ meters*

in graphene



*W. Stacklies, et al, PLoS Comp Biol, 2009
Costescu et al, BMC Biophys, 2012*

Forces from Molecular Dynamics simulations



1.

$$\begin{aligned}
 E = & \sum_{\text{bonds}} \frac{k_i}{2} (l_i - l_{i,0})^2 \\
 & + \sum_{\text{angles}} \frac{k_i}{2} (\theta_i - \theta_{i,0})^2 \\
 & + \sum_{\text{torsions}} \frac{V_n}{2} (1 + \cos(n\omega - \gamma)) \\
 & + \sum_{i=1}^N \sum_{j=i+1}^N \left(4\epsilon_{ij} \left(\left(\frac{\sigma_{ij}}{r_{ij}} \right)^{12} - \left(\frac{\sigma_{ij}}{r_{ij}} \right)^6 \right) + \left(\frac{q_i q_j}{4\pi\epsilon_0 r_{ij}} \right) \right)
 \end{aligned}$$

} bonded interactions \longleftrightarrow
 \longleftrightarrow
 } non-bonded interactions

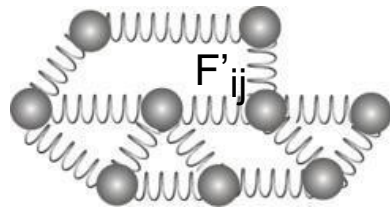
2.

&

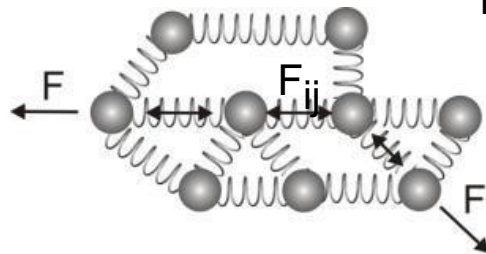
$$F_i = \frac{\delta E}{\delta r_i} = m_i \frac{d^2 r_i}{dt^2}$$

Forces from Molecular Dynamics simulations

F'_{ij} force between atom i and j in relaxed state



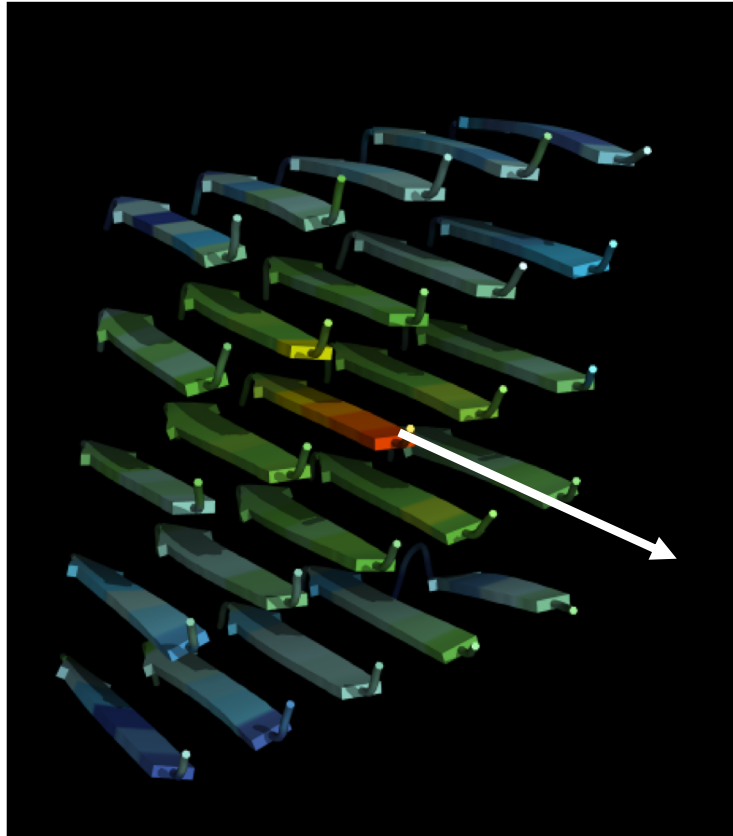
F_{ij} force between atom i and j in stretched state



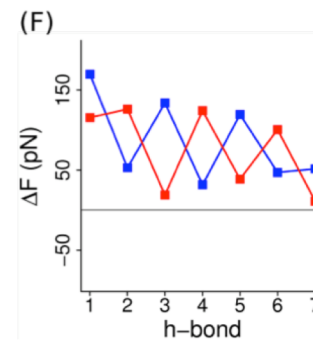
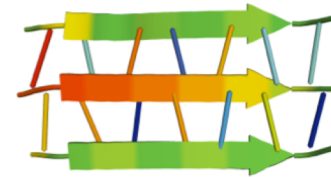
change in pairwise forces

$$\Delta F_{ij} = F_{ij} - F'_{ij}$$

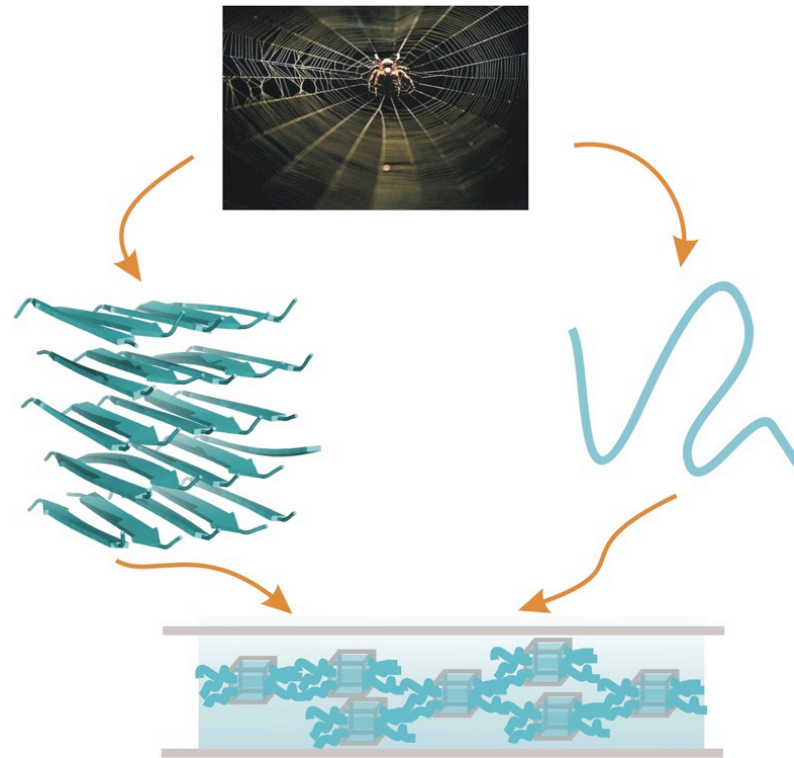
Mechanics of silk fibers



zigzag pattern of hydrogen bonds crucial for stabilization

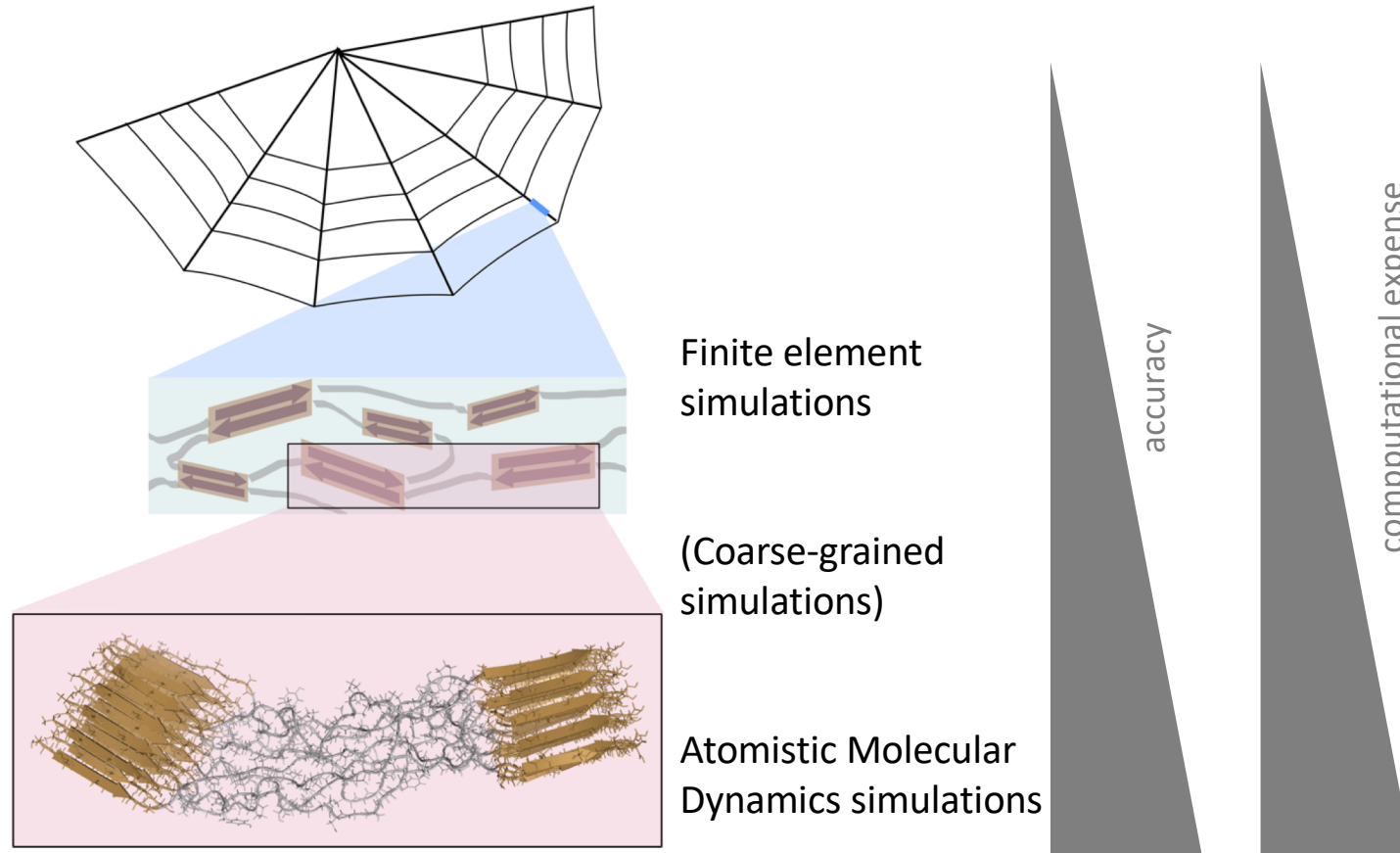


How to predict mechanics on whole fiber scale?

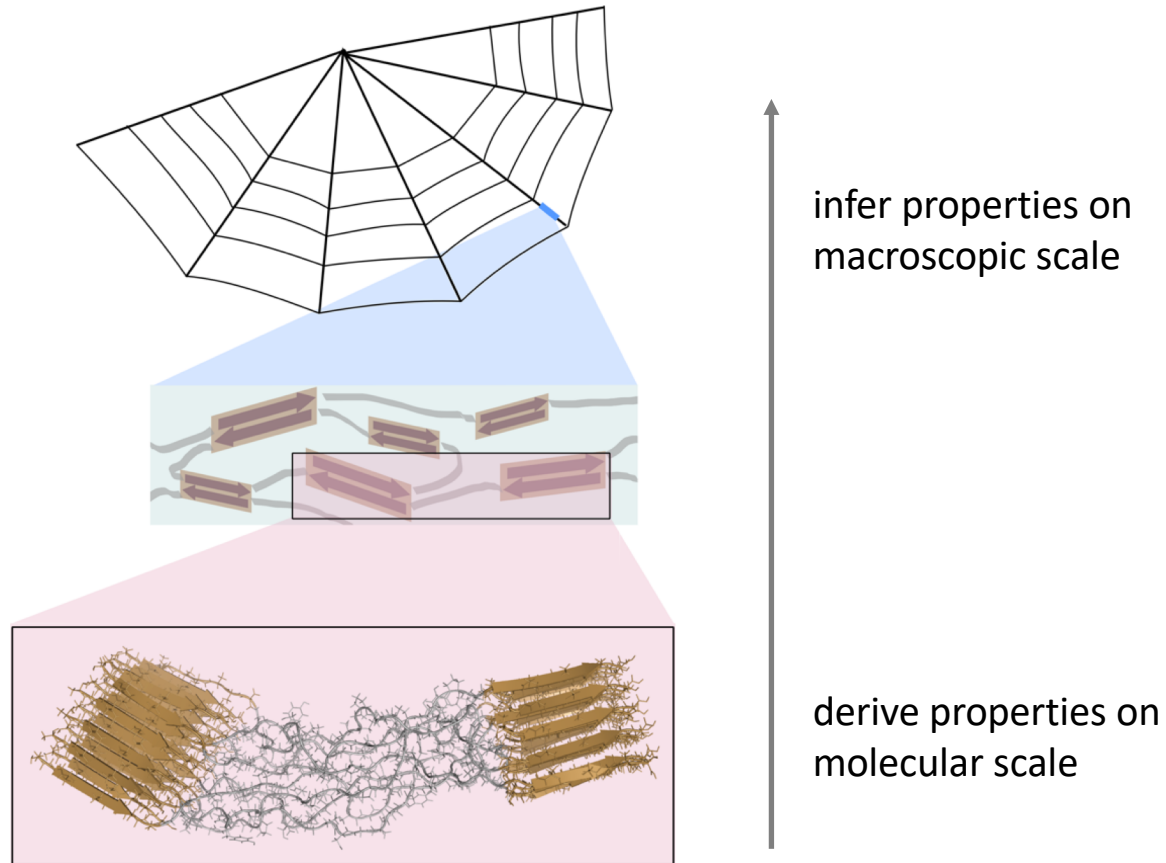


All-atom will not work....

Our method: bottom-up multiscale simulations

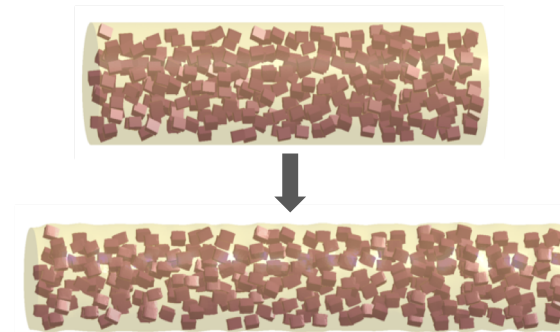


Our method: bottom-up multiscale simulations



Refined bottom-up finite element model of silk fibers

- structural parameters from SAXS, SANS et
- mechanical parameters from a large set of Molecular Dynamics simulations: elastic moduli, shearing moduli, Poisson's ratio
- varied to test robustness
- validation by comparison with macroscopic silk mechanics



Xiao et al, BJ 2009

Cetinkaya et al, BJ 2011

Cetinkaya et al, PCCP 2011

Patil et al, PLoS One 2014

Patil et al, BJ 2014

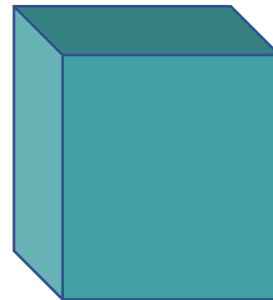
Wagner et al, Sci Rep 2017

Refined bottom-up finite element model of silk fibers

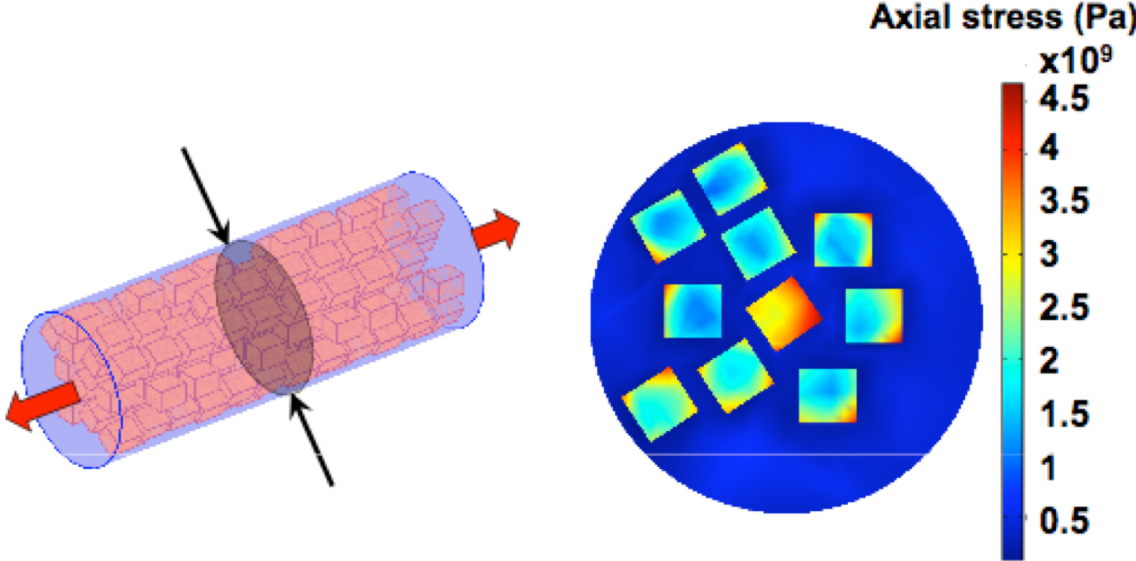
bear in mind: this complex structure...



...now is nothing but a cube

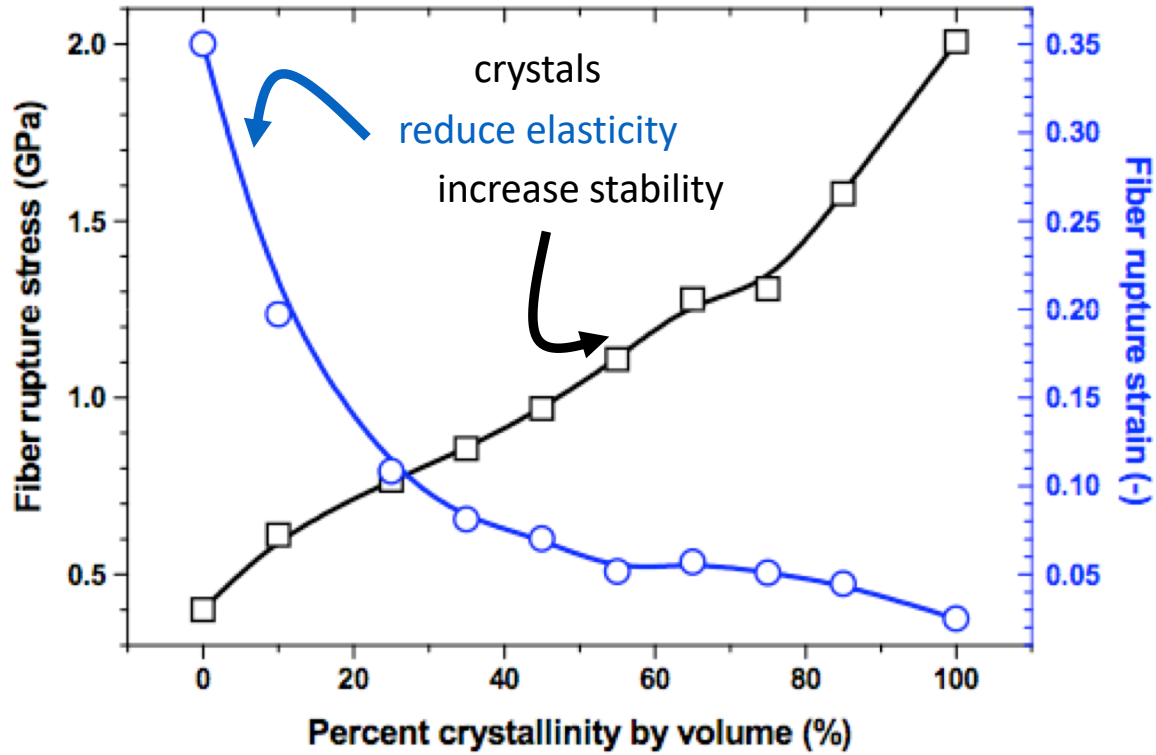


Our first simple finite element model of silk



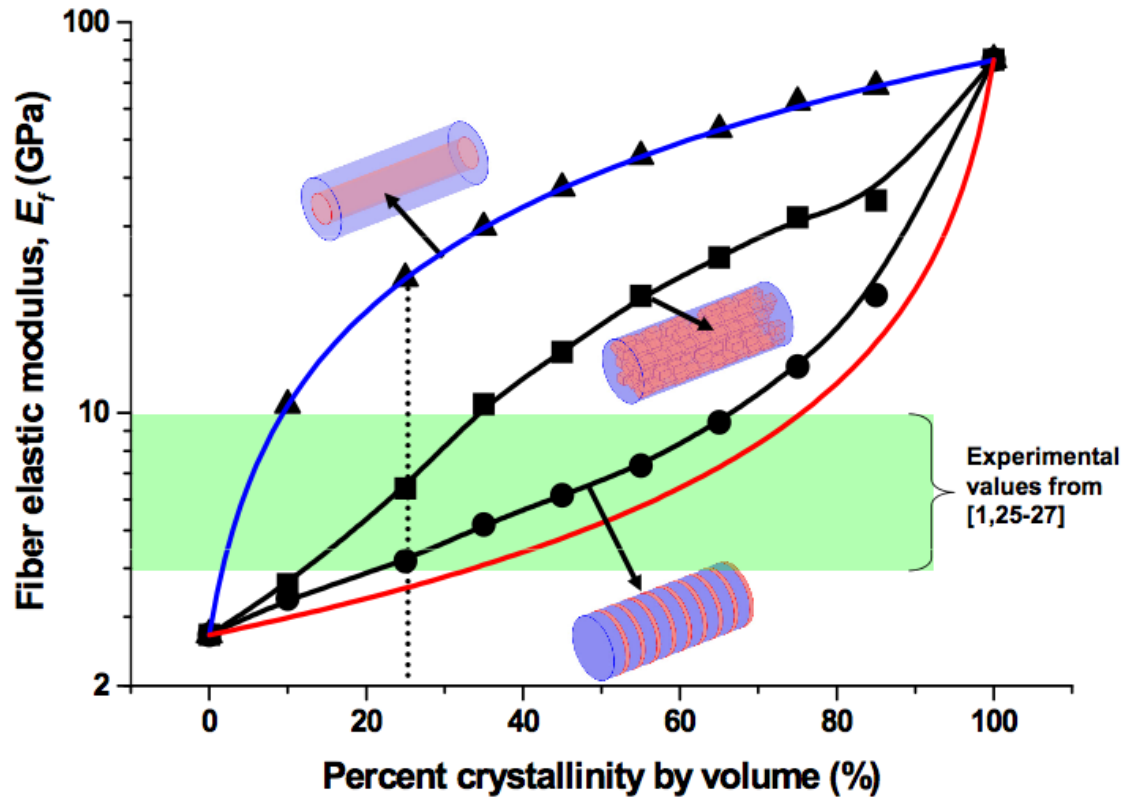
Our first simple finite element model of silk

stress =
force/area



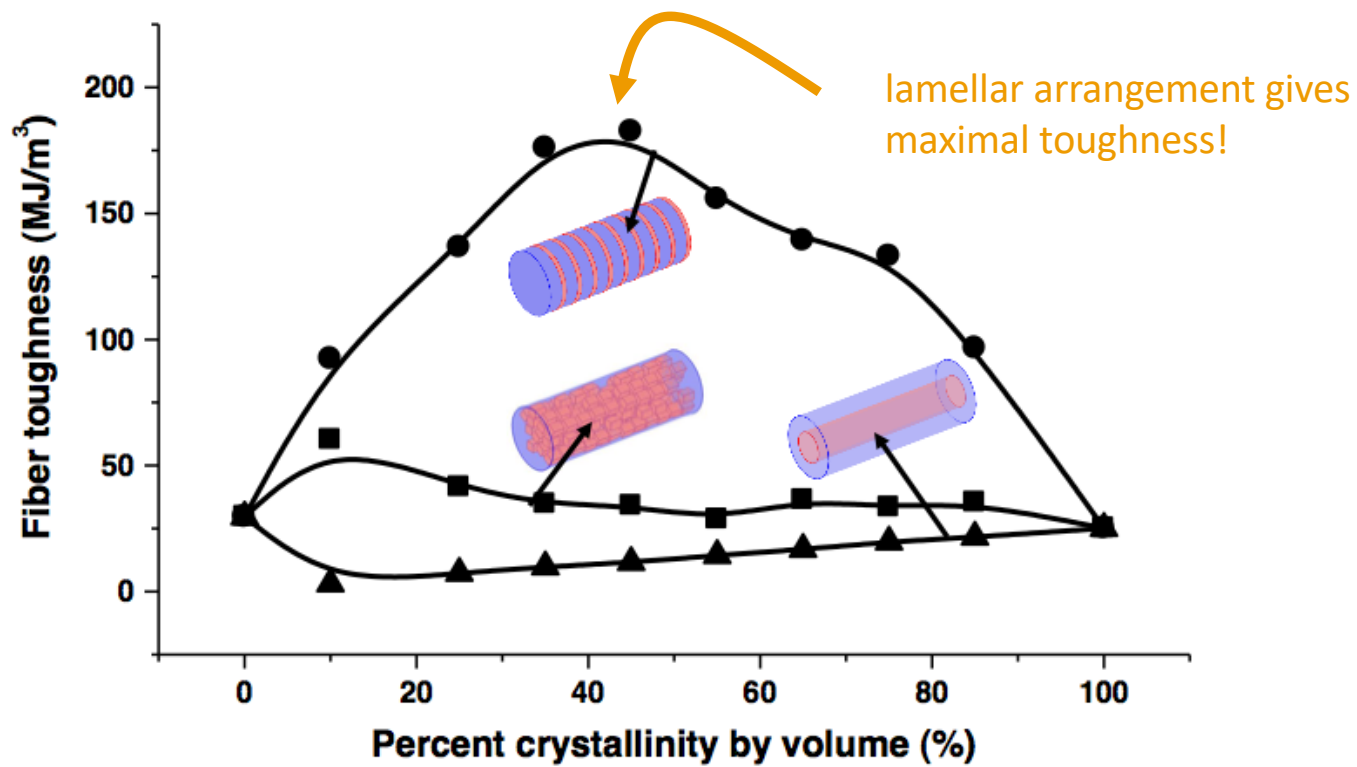
strain =
elongation relative
to initial length

Our first simple finite element model of silk

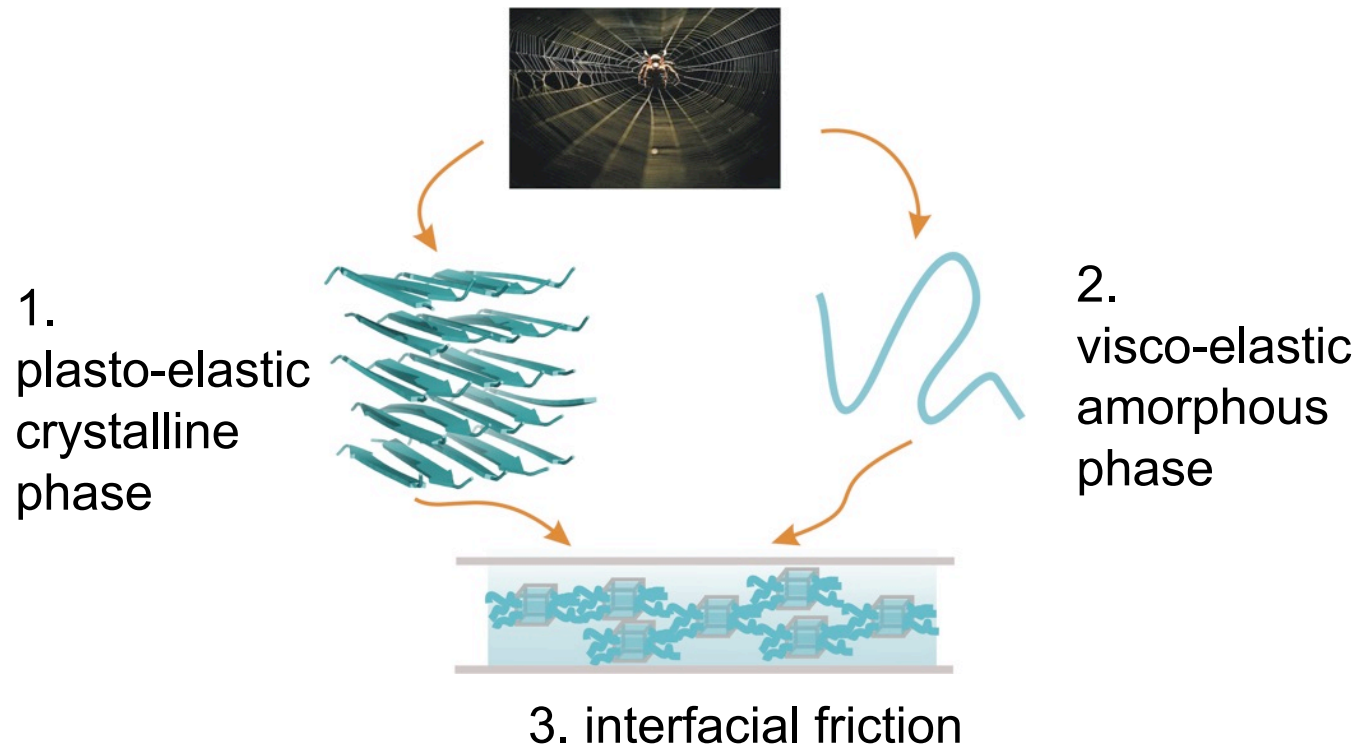


Toughness depends on long-range order

toughness =
energy
absorbed until
rupture
OR
integral under
stress-strain
curve



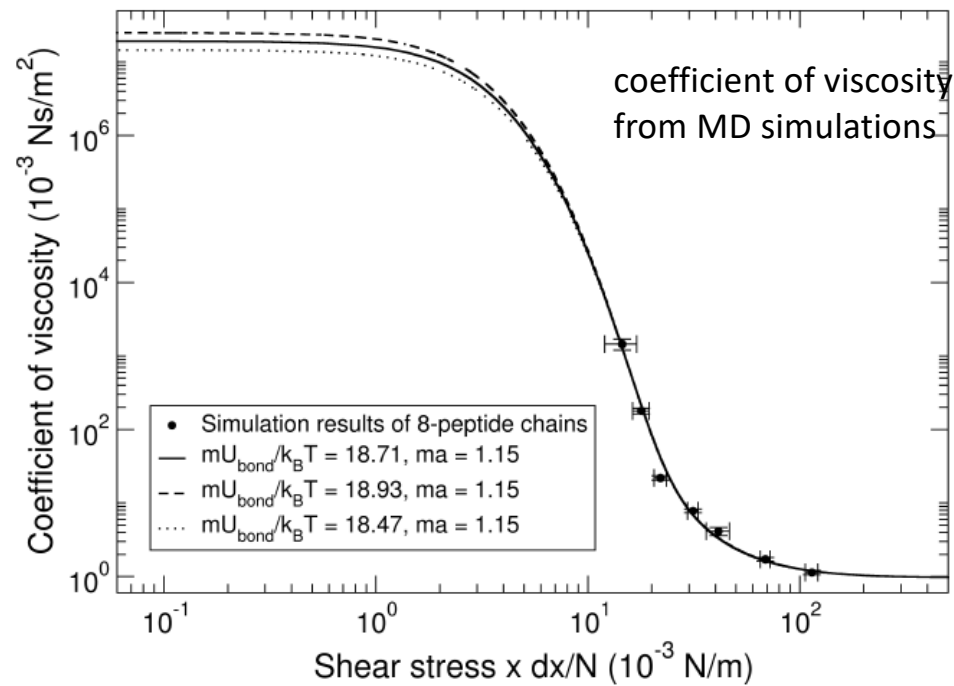
Next step: towards more realistic visco-elastoplastic silk



Next step: towards more realistic visco-elastoplastic silk

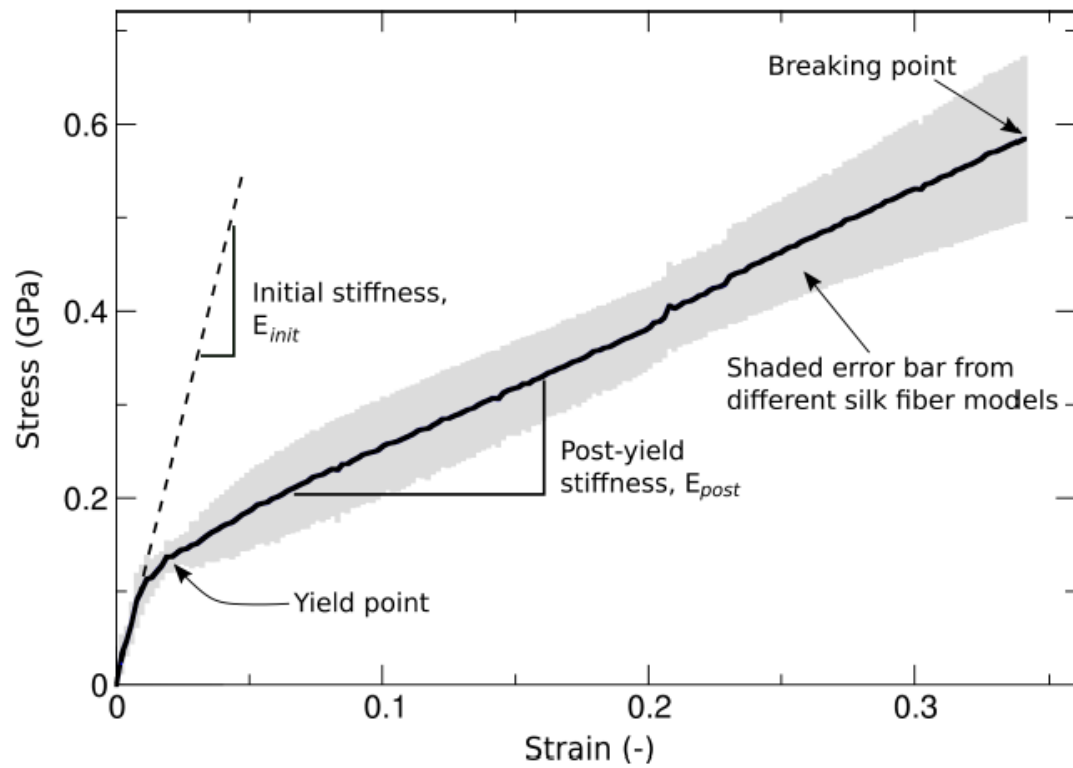


2. viscous
amorphous phase



Patil et al, BJ 2014; Patil et al, PLoS One

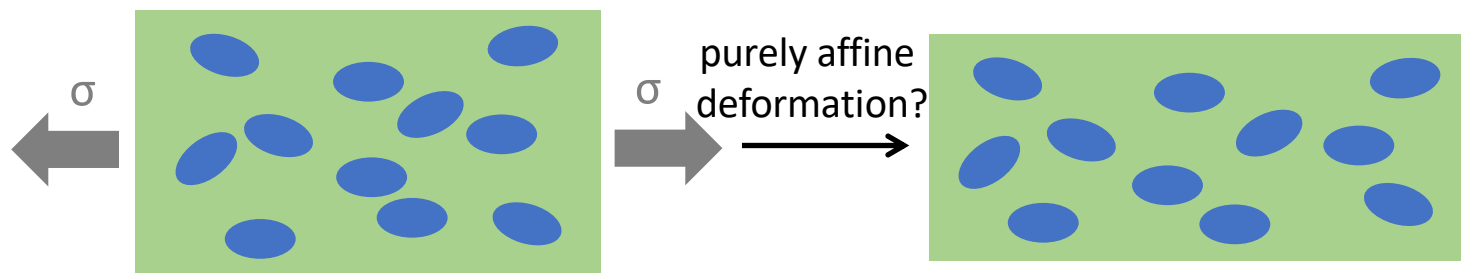
Next step: towards more realistic visco-elastoplastic silk



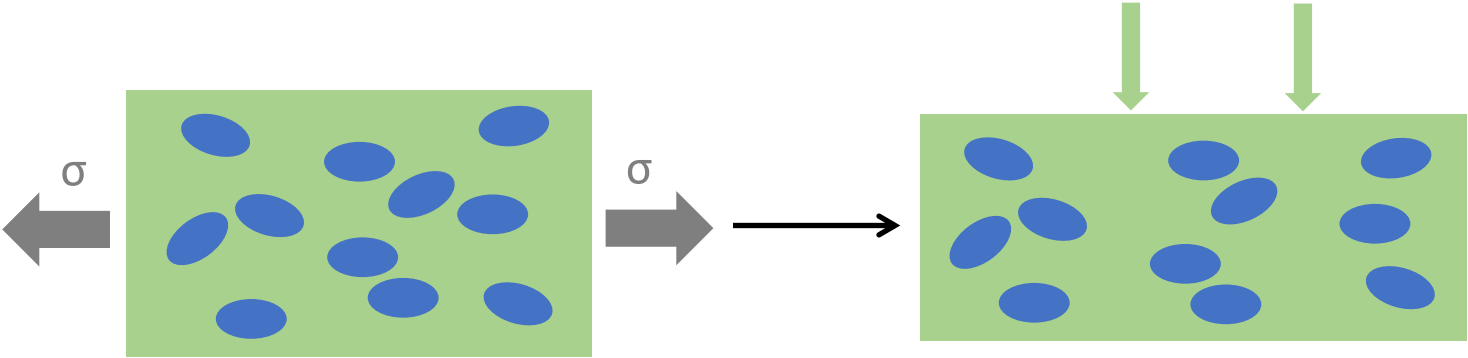
Next step: towards more realistic visco-elastoplastic silk

	Our Simulations results	Experiments by Denny(1976) [9]	Experiments by Gosline(1999) [3]
Ultimate Strength(GPa)	0.5-0.671	0.7-1.3	1.08-1.61
Extensibility	0.32-0.36	0.23-0.45	0.27-0.40
Initial stiffness or	6.5-8.1	3.8-9.8	7.0-10.0
Post-yield stiffness(GPa)	1.1-1.7	1.5-5.1	N/R
Toughness(MJ/m ³)	101-135	160	120-180
Hysteresis:	~70%	similar to experimental values for these strain rates	

Using this new model: self-ordering in spider silk fibers



Using this new model: self-ordering in spider silk fibers

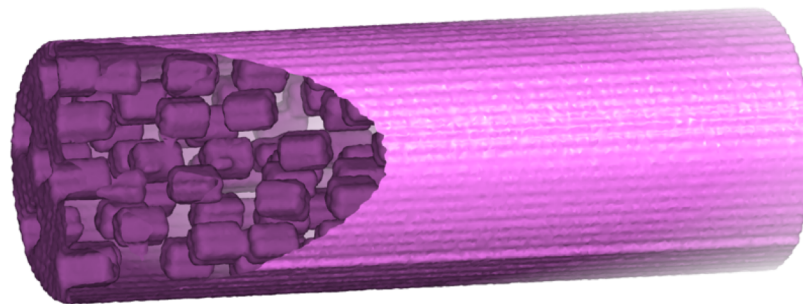


Finite element model

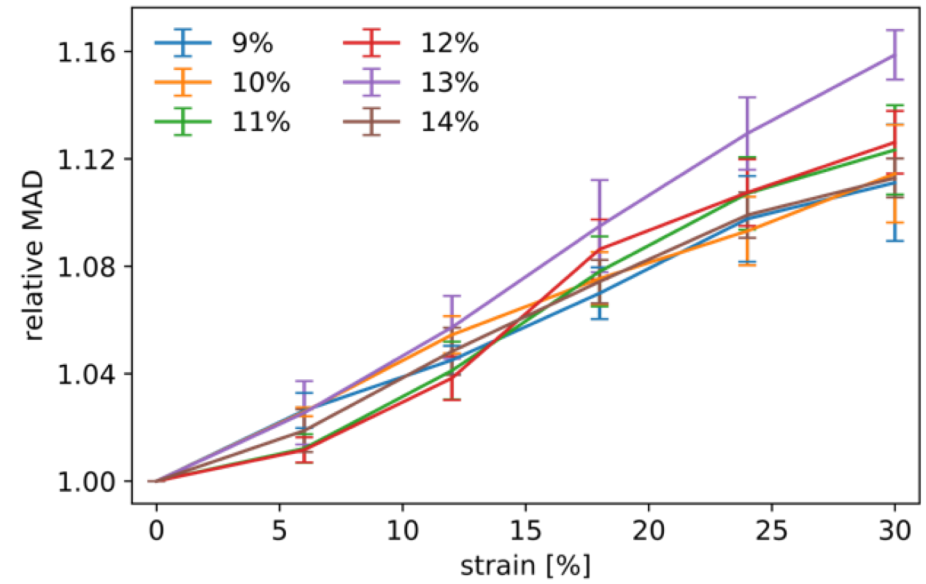
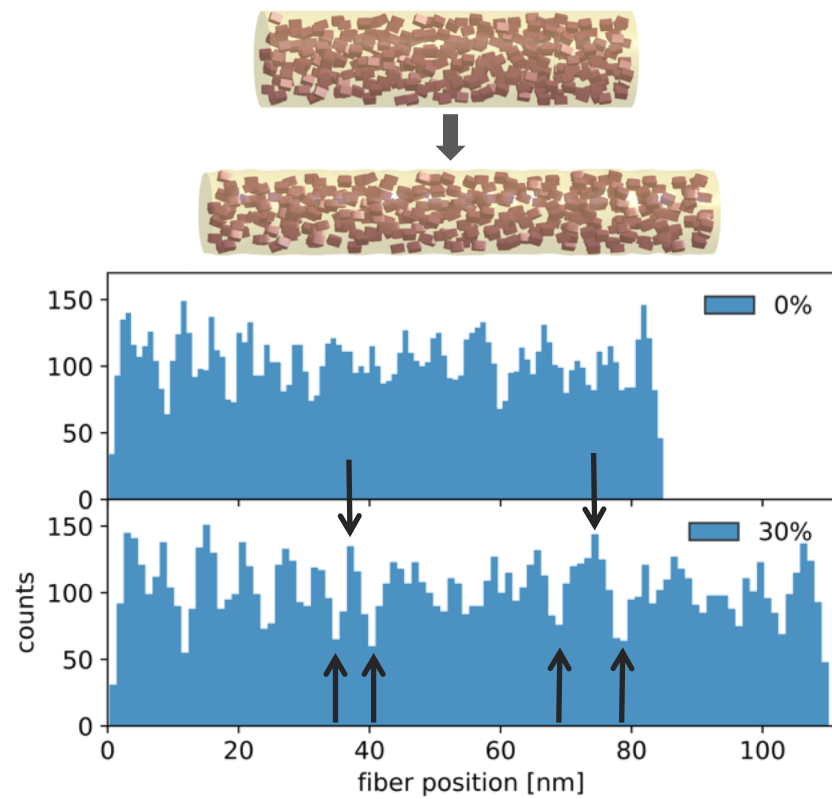
cubes represent beta-sheet crystals, sizes/angles from scattering experiments



surrounded by soft amorphous matrix

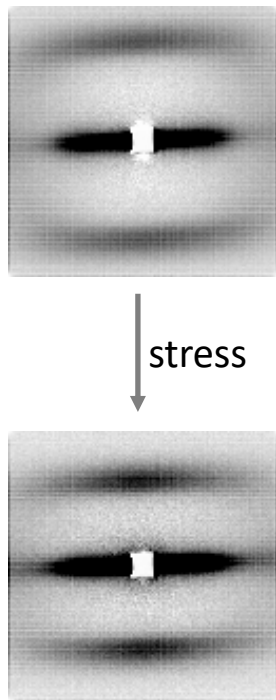


Stretch-induced self-ordering: computational prediction

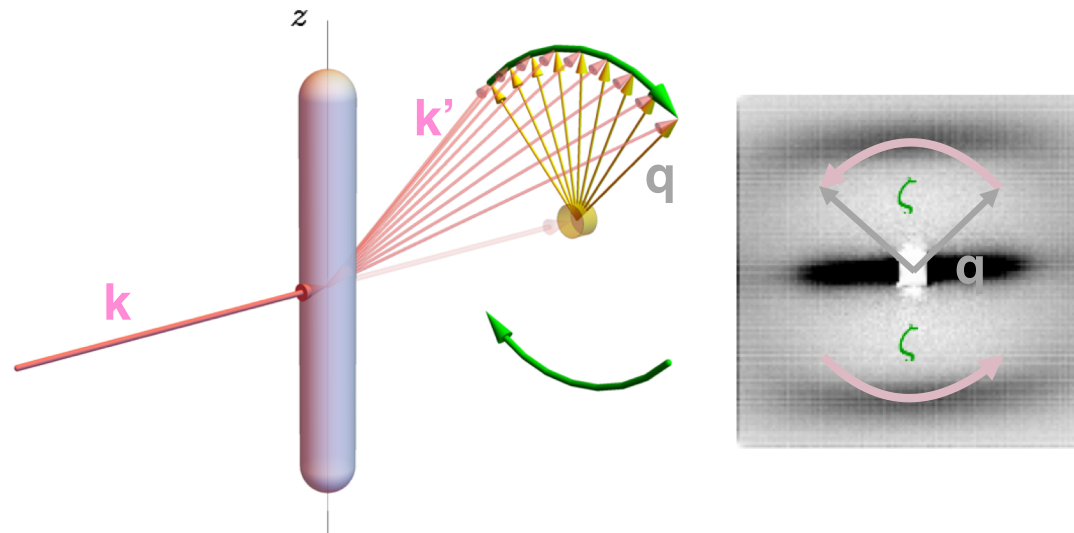


BUT: is this real? Any experiments to test this?

Stretch-induced self-ordering: confirmed by small angle neutron scattering



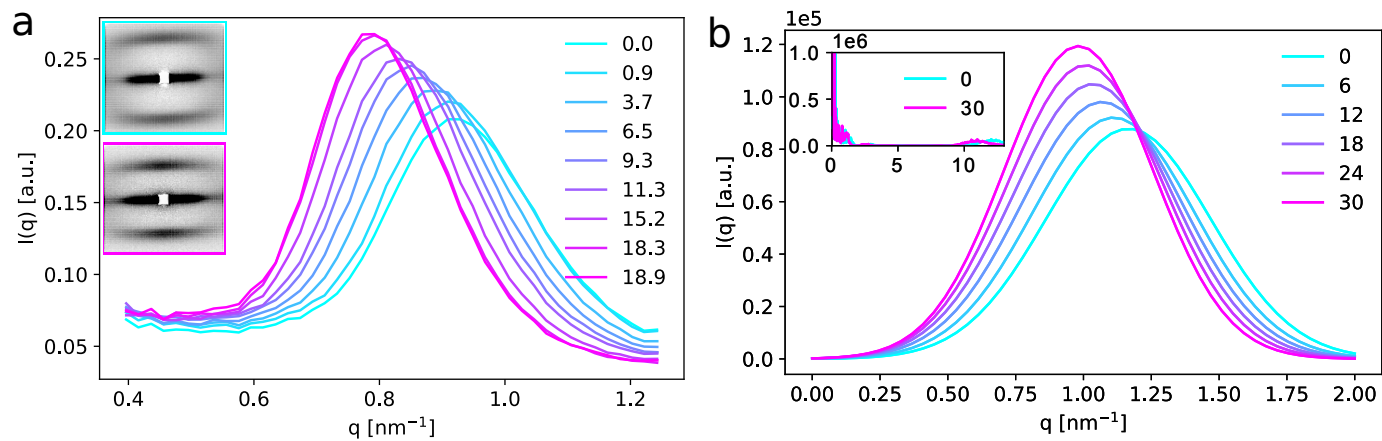
- angular integration in $\zeta \in \left[-\frac{\pi}{4}, \frac{\pi}{4}\right]$



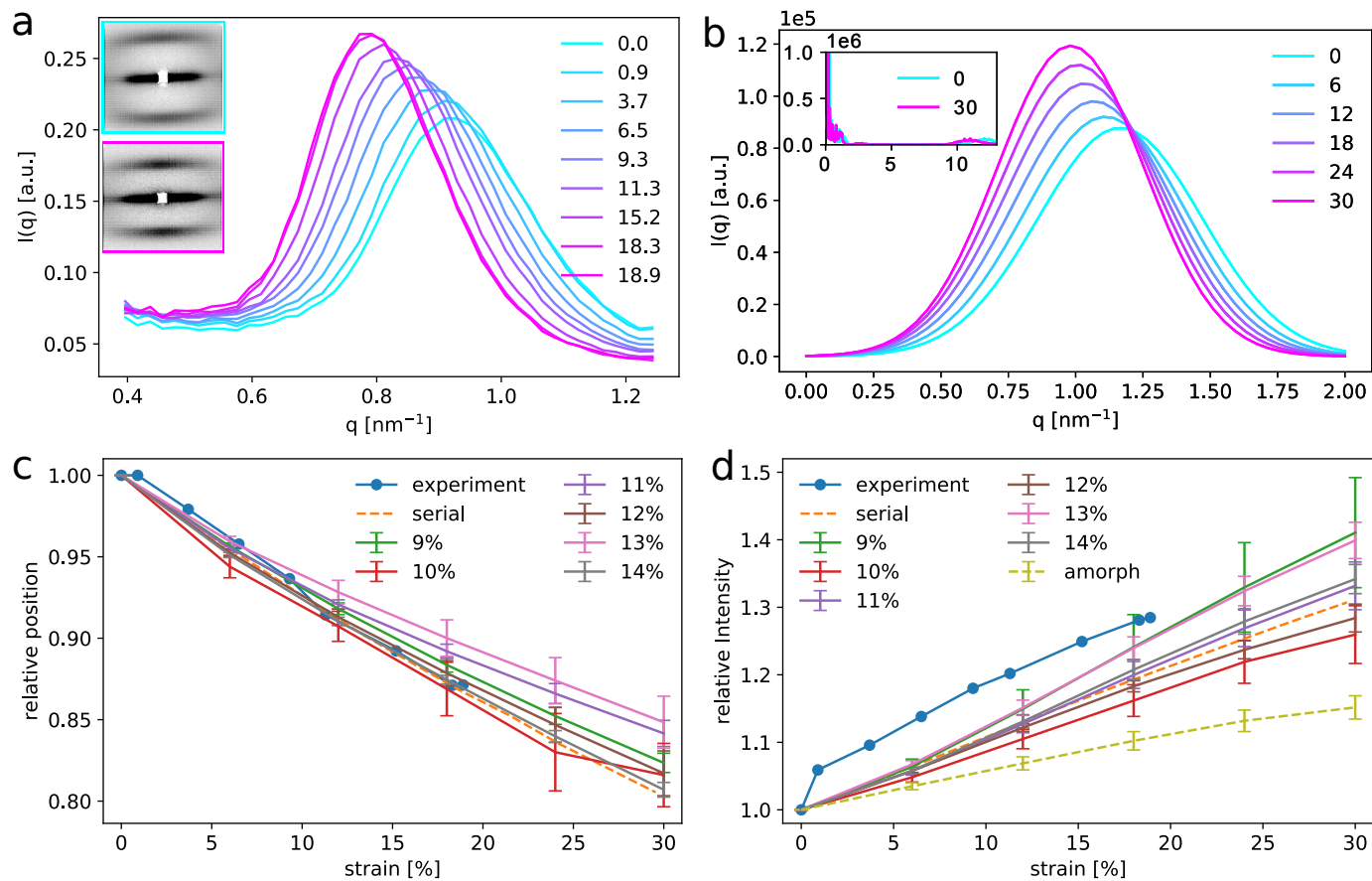
- apply rotation matrix on \mathbf{q} and sum over ζ :

$$S(\mathbf{q}) = \frac{1}{N} \sum_{\zeta} \sum_{i,j} e^{-i\langle \mathbf{R}(\zeta)\mathbf{q}, \mathbf{r}_{ij} \rangle}$$

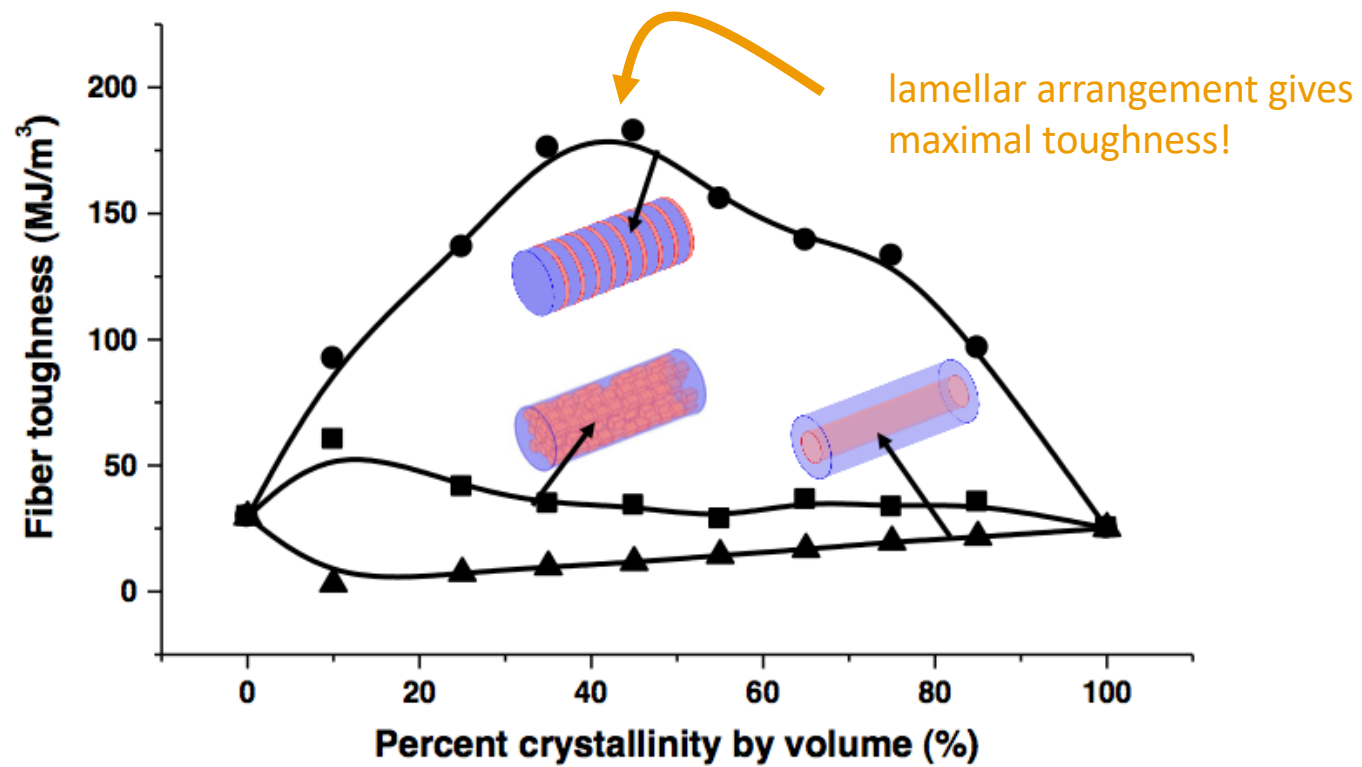
Stretch-induced self-ordering: confirmed by small angle neutron scattering



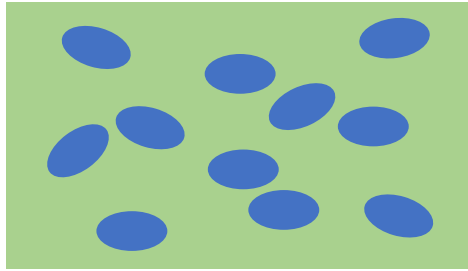
Stretch-induced self-ordering: confirmed by small angle neutron scattering



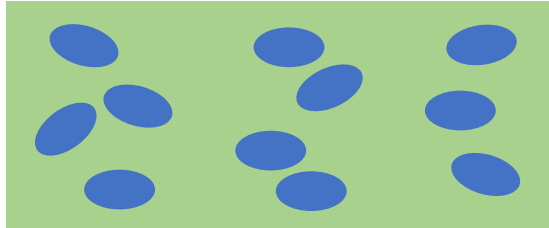
Toughness depends on long-range order



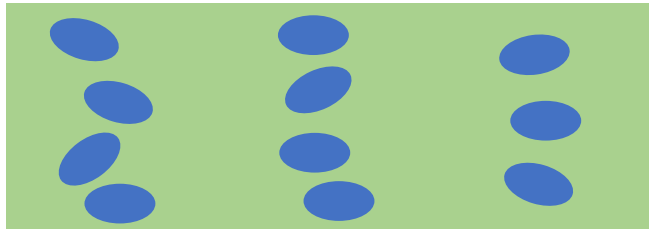
Stretch-induced self-ordering



random
distribution



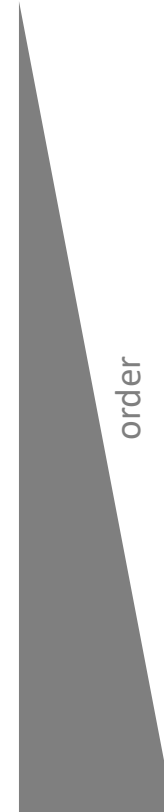
partial
order



high
order

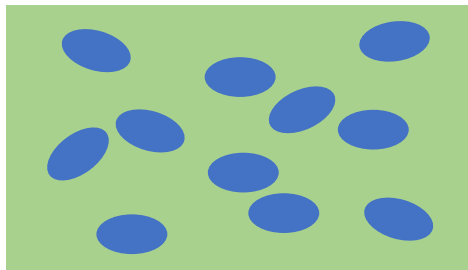


stretching



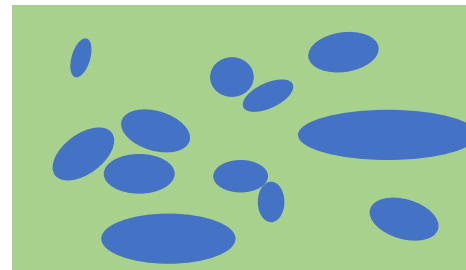
Stretch-induced self-ordering: silk as a block copolymer

silk or block-copolymer



crystallizes into similarly-sized nanoparticles from (Ala)_x

semi-crystalline polymer (e.g. polystyrene)

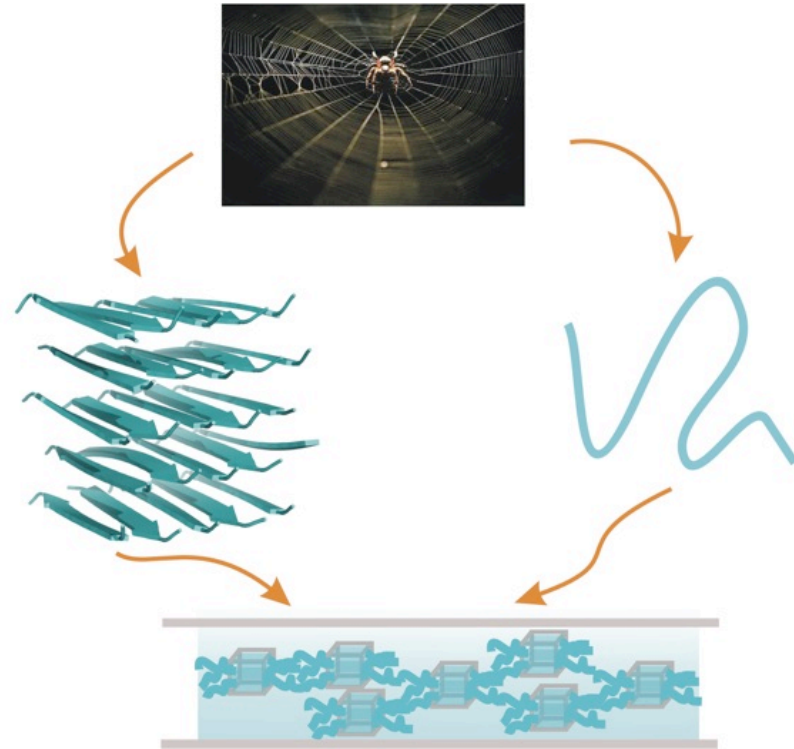


crystallizes here and there and everywhere

Conclusions

MD simulations & Finite element analysis can in combination bridge scales

Quantitative predictions by simulations? Yes!





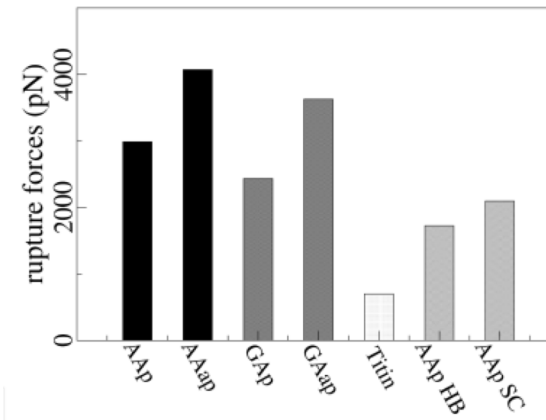
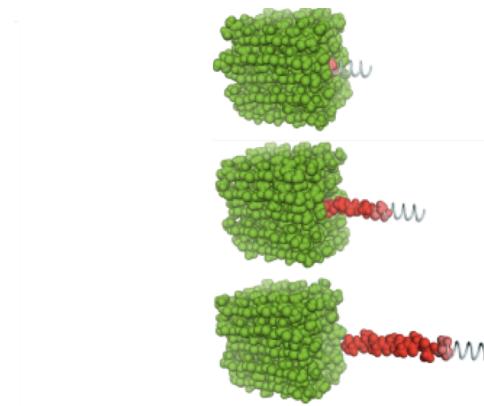
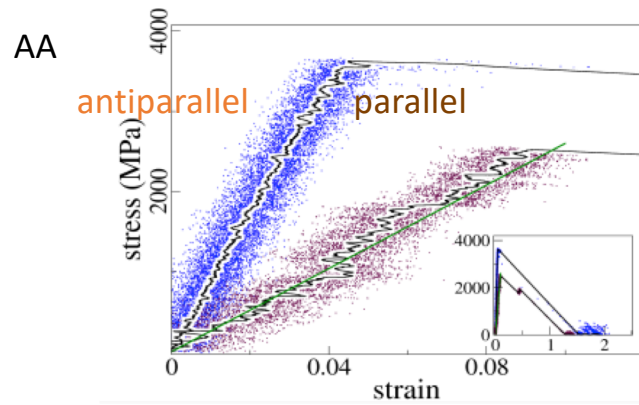
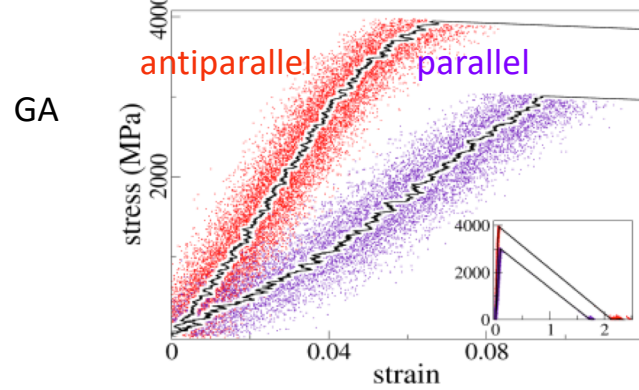
Molecular Biomechanics

Ana Herrera-Rodriguez
Nicholas Maragakis
Florian Franz
Fabian Kutzki
Christopher Zapp
Agnieszka Obarska-Kosinski
Fan Jin
Benedikt Rennekamp
Isabel Martin
Anna Schroeder

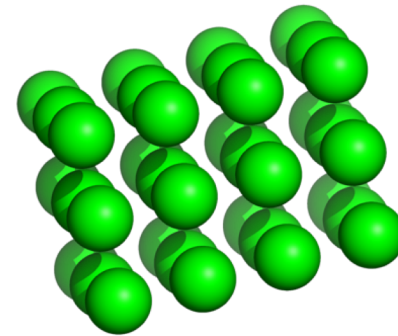
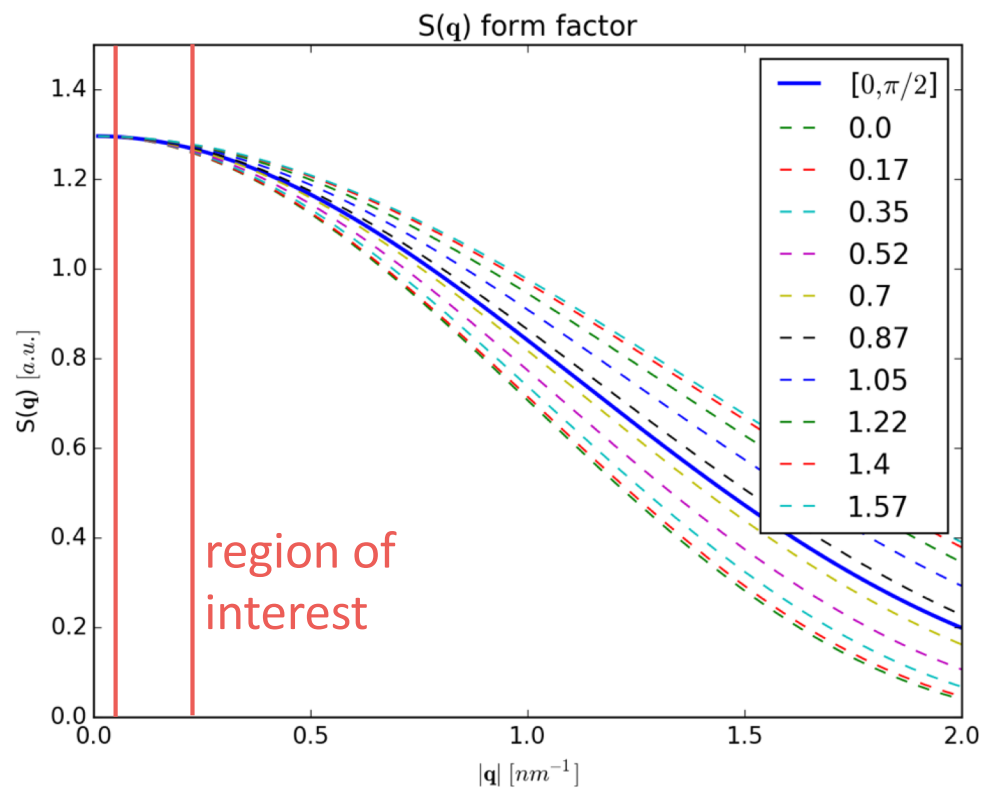
\$\$:
Klaus Tschira foundation
DFG, Volkswagen Foundation,
AvH,
Toyota

Mechanics of silk fibers

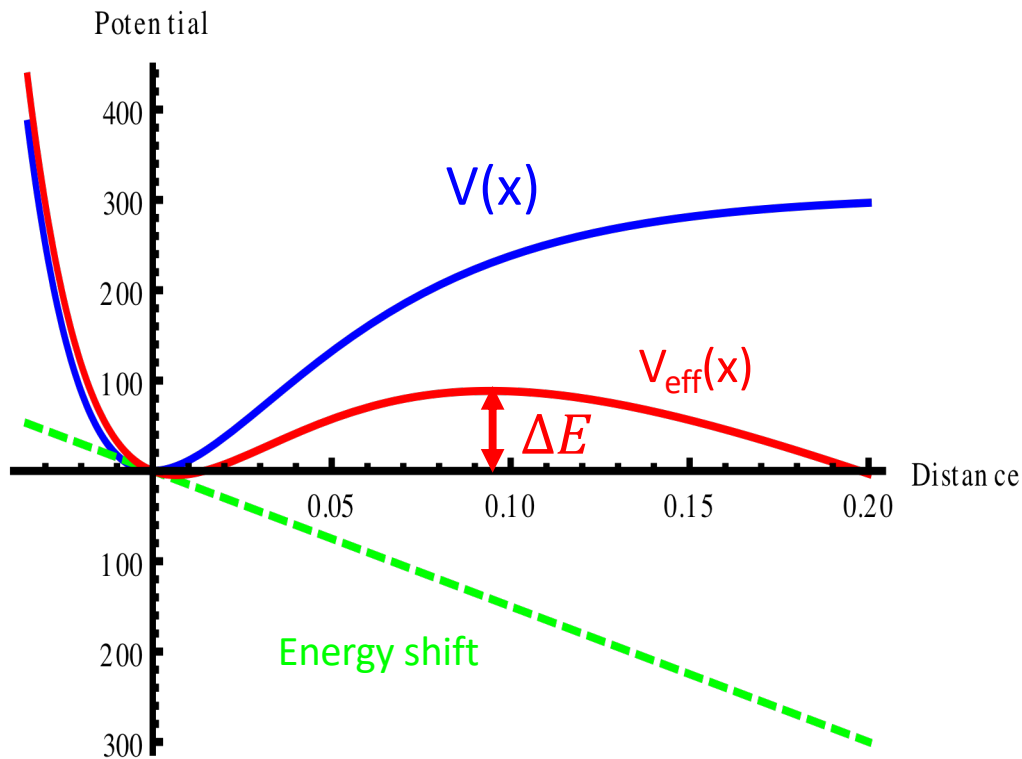
stress-strain curve
from Molecular Dynamics



Form factor only



Model: How to calculate rupture rates?



Using Bell-Evans model for the potential and Arrhenius kinetics to calculate rupture rates; taking parameters for the potential from force field.

$$V(x) = D[1 - \exp(\beta(x - x_0))]^2$$

$$V_{eff} = V(x) - x \cdot V'(x_{curr})$$

$$\Delta E = V_{eff}(x_{max}) - V_{eff}(x_{min})$$

$$\Rightarrow k_{rupture} = A \cdot \exp(-\Delta E/k_B T)$$